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QUARTERLY REPORT  
MICROELECTRONICS RESEARCH ON  
SILICON-SILICON OXIDE  
STRUCTURES AND INTERFACES  
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NAS 8-11235  
Quarterly Report Summary  
Microelectronics Research on Silicon-  
Silicon Oxide Structures and Interfaces

Summary

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The development of capability towards satisfactory completion of the program continues. Some of the considerable recently-published information on M-O-S structures is reviewed. A major point of agreement on space charge in the oxide has been reached. Significant details are yet to be explained.

The experimental program now has operating the process furnace, the electrolytic etching fixture, the capacitance bridge, the replication evaporator, and the electron microscope. The deionized water system produces 18 meg ohm-centimeter water. Simple usage of photolithographic techniques has been satisfactorily demonstrated. The necessary laboratory operations can now be executed.

Fluorescence techniques appear to offer a sensitive analytical tool for measuring the organic content in deionized water supplies. Investigations to evaluate the utility of the techniques can be performed with additional equipment. Electron mirror microscopy appears to need further development to have the resolution desired for cross section investigations. For studying the uniformity and structure of oxide surfaces this tool could be quite useful.

New general bibliography sections cover report literature recently surveyed.



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MICROELECTRONICS RESEARCH ON  
SILICON-SILICON OXIDE STRUCTURES AND INTERFACES

I. Models of Oxide Structures

A. Physical Model of the M-O-S Device

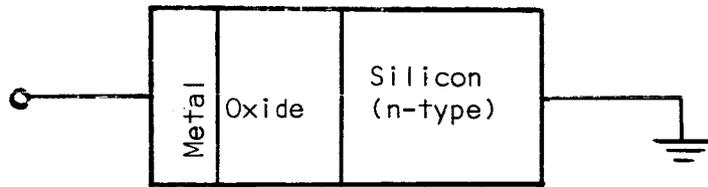


Figure 1  
Cross Section of M-O-S Structure

Consider the case when the metal is biased positive with respect to the n-type semiconductor. There will be positive charges at the metal-oxide interface and electrons at the silicon-silicon oxide interface. The result is that there will be an accumulation region in the silicon (particularly at the silicon-oxide interface).

The measured capacitance will be that of the oxide and is given by the expression

$$C_0 = \frac{\epsilon_1 S}{X_0} \quad (1)$$

where  $\epsilon_1$  is the permittivity of the oxide,  $S$  is the cross sectional area of the device, and  $X_0$  is the width of the oxide layer.

Next consider the case when the metal is biased negative with respect to the semiconductor. Electrons will be transported from the semiconductor to the metal leaving exposed impurity ions in the semicon-

ductor thus creating a space charge region. By the condition of charge neutrality the total charge in the depletion region is equal to the charge on the metal plate. This value is given by

$$Q = CV \quad (2)$$

where C is the measured capacitance of the device and V is the voltage drop across it.

This is equivalent to a parallel plate capacitor with two dielectrics between the plates as in Figure 2 where  $d_1$  is the dielectric number one, the oxide, and  $d_2$  is the dielectric number two, the semiconductor.

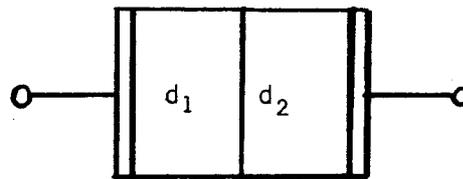


Figure 2  
Two Dielectric Capacitor

The capacitance for this configuration is

$$C = \frac{C_1 C_2}{C_1 + C_2} \quad (3)$$

where

$$C_1 = \frac{\epsilon_1 S}{X_1}, \quad C_2 = \frac{\epsilon_2 S}{(X_2 - X_1)}$$

The M-O-S device can therefore be represented as two capacitors in series as in Figure 3. One capacitor is due to the oxide layer and is constant; the other is due to the depletion region and is a function of bias.

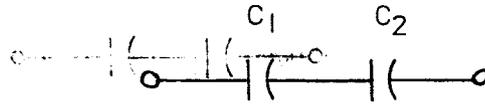


Figure 3  
Capacitors in Series

For the purpose of this derivation it will be assumed that the capacitance vs. voltage curve for the device is available from measurements made on the device. The capacitance will be the series equivalent of the two capacitors and the voltage will be the total barrier voltage. The capacitance due to the depletion region can be found from the expression

$$C_d = \frac{C_o C}{C_o - C} \quad (4)$$

The total voltage is the sum of the drops across the two capacitors, the contact potential, and the IR drop

$$V_T = V_o + V_d + V_c + IR \quad (5)$$

where

$V_T$  = applied voltage

$V_o$  = oxide voltage drop

$V_d$  = depletion layer voltage drop

$V_c$  = contact potential

$IR$  = drop due to leakage current.

The leakage current  $I$  will be measured but its value will probably be such that it can be neglected. It is hoped that the contact potential can be determined during the course of this work; however, for the present this

will be neglected or a constant assumed (on the order of 0.5 volts). The simplified voltage is now

$$V_T = V_O + V_d \quad (6)$$

The voltage drop can be determined in the following manner:

$$Q_T = Q_O = Q_d \quad (7)$$

This is equivalent to saying that the charge is the same on all plates of the capacitor.

$$CV = C_O V_O = C_d V_d \quad (8)$$

$$\frac{C_O C_d}{C_O + C_d} (V_O + V_d) = C_O V_O = C_d V_d$$

$$V_d = \frac{C_O V}{C_O + C_d} \quad (9)$$

Thus, the capacitance of the depletion region and the voltage drop across it can be determined from the voltage vs. capacitance curve.

If a small reverse bias is applied to the device the depletion region formed will be narrow. For this region the charge density can be assumed to be constant. The plot of charge density vs. distance for this condition is shown in Figure 5.  $x_{d1}$  is the depletion width of this small bias.

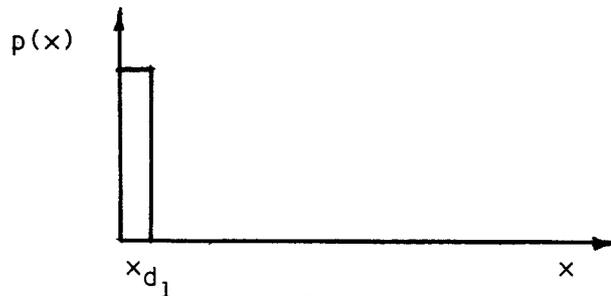


Figure 4a  
Charge Density Versus Distance

Assuming the charge density varies only with X (one dimensional case), the relationship can be expressed as

$$Q_{d_1} = p_1(x)X_{d_1} \quad (10)$$

thus, both the charge density and the depletion width can be obtained from the C vs. V curve.

Next the bias is increased by a small amount. This results in more charge being uncovered and the widening of the depletion region.

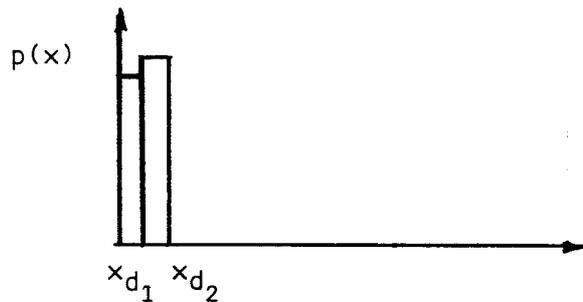


Figure 4b  
Charge Density Versus Distance

The situation might be that of Figure 6. The depletion width is given by

$$C_{d_2} = \frac{\epsilon S}{X_{d_2}} \quad (11)$$

and the charge is given by

$$Q_{d_2} = C_{d_2} V_{d_2} = p_1(x)X_{d_1} + p_2(x)(x_{d_2} - x_{d_1}) \quad (12)$$

thus, the charge density can be determined as well as the depletion region.

The bias is now increased slightly and the process is repeated. This is done until a complete curve of charge density vs. distance is obtained.

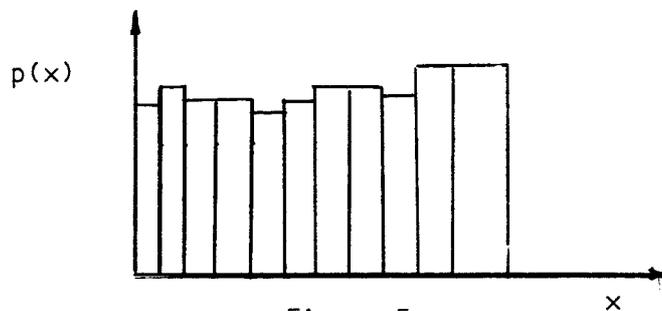


Figure 5  
Histogram of Charge Density Versus Distance

In a depletion region the charge density is given by

$$p(x) = e[N_d - N_a] \quad (13)$$

where

$e$  = charge on the electron

$N_d$  = donor impurity concentration

$N_a$  = Acceptor impurity concentration

Thus, the impurity profile can be calculated from the charge density curve.

#### B. Recent Publications on M-O-S Structures

During the last several months there have been several papers published on the M-O-S device. A large number of these papers have been written by scientists with the IBM Corporation<sup>1</sup> and with Fairchild Semiconductor.<sup>2</sup> One of the major results of their studies has been to determine that the ions can migrate throughout the oxide of the M-O-S device and create a space charge in the oxide. The ions are thought to be oxygen vacancies in the case of the Si-SiO<sub>2</sub> system. Use of a model based on oxygen ion vacancy migration can be used to explain some of the major phenomena of M-O-S devices while other phenomena can be explained only with a surface state model. Other phenomena are yet to be explained. At the present understanding of the M-O-S device, a model which uses both the

surface state and ion migration phenomena gives the most complete picture. A review of the models follows.

Due to the abrupt change from a periodic crystal lattice at the surface of a semiconductor, energy states exist for carriers that are not allowed in the bulk of the material. These states in the surface state model may trap carriers and as a consequence the surface can become charged. A method for determining the charges trapped in surface states is shown below in Figure 6.

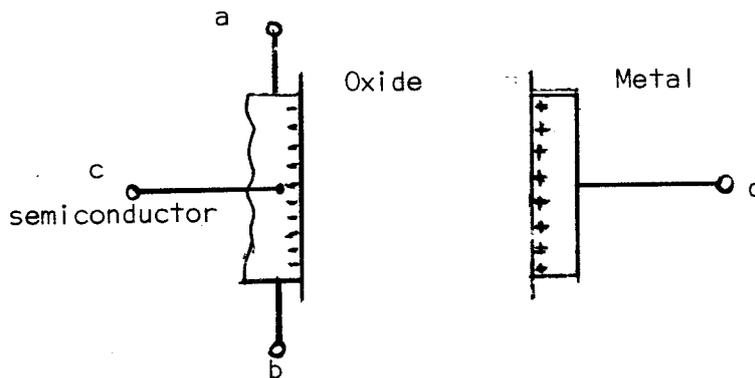


Figure 6  
Conductance Electrode M-O-S Structure

The bias between points c and d can be varied thus changing the charge at the metal oxide interface and at the semiconductor oxide interface. The charge at the semiconductor-oxide interface will result in a change in the conduction between points a and b. If the mobility of the charge carriers is known, the theoretical change in conduction between points a and b may be calculated. In practice the change in conductivity is not usually the theoretical value but lower and a function of frequency. The explanation most used is that of the surface states. If a low frequency

signal is used to modulate the conductivity between a and b, the states are filled and emptied as the signal varies. Since the situation is that of steady state the maximum number of surface states would be filled (maximum number of carriers trapped) and the conductivity change between "a" and "b" would be a minimum.

If the frequency of the signal between "c" and "d" is increased, some of the states will not be filled due to insufficient time. In this case there will be more carriers to conduct current between "a" and "b" and the change in conductivity between these points would be higher than in the previous case. If the frequency is further increased, the conductivity change will increase until at some high frequency none of the surface states will be filled and the change in conductivity between "a" and "b" will be a maximum for the applied signal between "c" and "d". Another way of demonstrating this effect would be to apply a negative step voltage at point "c". The conductivity would change immediately by an amount corresponding to there being no surface states but would then decay exponentially as the surface states were filled to some value corresponding to the situation of all of the surface states being filled. This is shown in Figure 7. The step voltage is applied at  $t = 0$ .

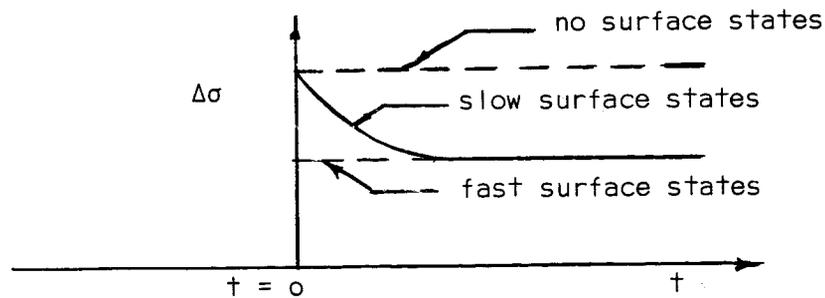


Figure 7  
Idealized Charge-Conductivity-Time Relationship

If the step voltage is left at "c" for an extended time the conductivity is noted to decrease still further. This would seem to correspond to filling of extremely slow surface states. The experiment may have to be conducted at elevated temperatures to observe this secondary effect but since the filling and emptying of surface states is a function of temperature, this is not unreasonable. However, when the step voltage is removed the conductivity changes to a value below its original value and the conductivity is stable at this point. This case is shown in Figure 8.

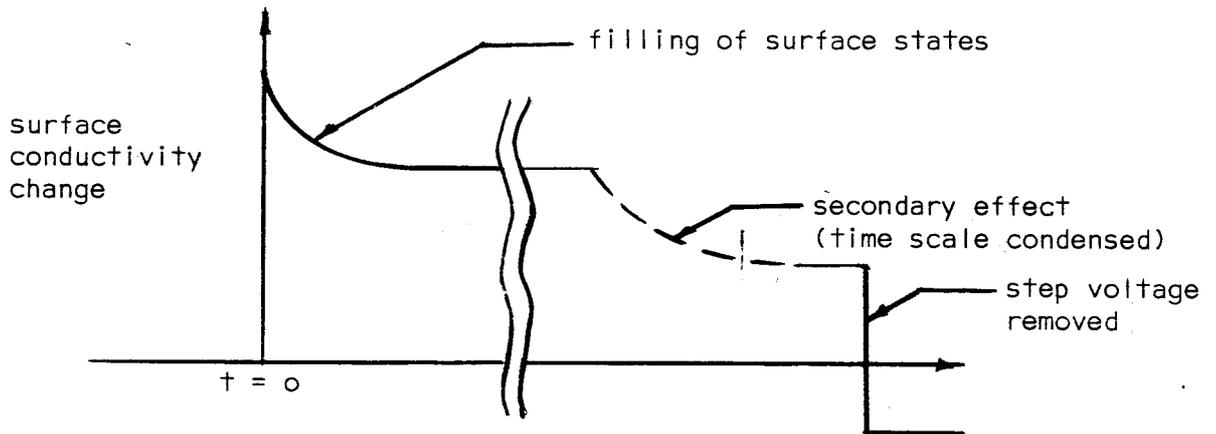


Figure 8  
Observed Charge-Conductivity-Time Relationship

The secondary effect has been explained in the following manner. During the time when the step voltage is applied to point "c", ions in the oxide layer migrate to the semiconductor-oxide interface (especially if the experiment is conducted at higher temperatures). These ions tend to pile up at the interface. The arrangement is semipermanent, thus creating a significant space charge field within the oxide. When the step voltage is

removed from point "c" the space charge remains.

In the case of the Si-SiO<sub>2</sub> system the ions in the oxide are thought to be oxygen vacancies. This tends to be supported by the fact that the space charge is usually positive and that the presence of an oxidizing agent tends to stabilize the effect while the presence of a reducing agent aggravates it. The oxidizing agent is thought to be a source of oxygen to fill the vacancies and the reducing agent is thought to capture some of the oxygen from the SiO<sub>2</sub>. In the experiments reported, aluminum was the metal used to make the contact with the silicon-dioxide and it is thought that the oxygen in the oxide can react with the aluminum thus creating the vacancies. The vacancies, once created, are able to migrate throughout the oxide under the influence of an electric field and/or temperature.

This model of contact-metal-caused vacancies can be used to explain some of the phenomena associated with the M-O-S structure. One of the phenomena is that the M-O-S structure can be considered as a diode with the ion current flow 100 times or more greater in one direction than when biased in the following direction.

If the metal is biased negative, oxygen ions will be forced to the Si-SiO<sub>2</sub> interface under the influence of the field. However, since the ions can only migrate if vacancies exist and the vacancies are assumed to be formed at the aluminum-oxide interface there will be little current flow due to the fact that the silicon-oxide is saturated a little ways from the metal. If the metal is now biased positive, oxygen ions will be attracted to it thus letting the vacancies migrate toward the silicon.

Since the oxygen is assumed to react with the metal, vacancies can be formed at the metal-oxide interface and the process can continue. The current will be much larger in this case. The space charge created with the metal biased positive will be large.

The space charge in the oxide can be studied by measuring the capacitance versus voltage of the M-O-S device. Theoretical curves of capacitance versus voltage can be calculated where the applied voltage  $V_{app}$  is equal to

$$V_{app} = \phi_{ms} + \frac{Q_{ss}}{C_o} \quad (14)$$

where  $\phi_{ms}$  is the work function difference between the metal and semiconductor and  $Q_{ss}$  is charge density of the surface states. If there is a space charge in the oxide, an additional term represents its effect on the potential.

$$V_{app} = \phi_{ms} + \frac{Q_{ss}}{C_o} + \frac{1}{\epsilon_o \epsilon_o} \int_0^{x_o} xp(x) dx \quad (15)$$

The term  $\int_0^{x_o} \frac{1}{\epsilon_o \epsilon_o} xp(x) dx$  corresponds to an effective charge located at the Si-SiO<sub>2</sub> interface. The terms represent a horizontal shift of the C versus V curves. If the C versus V curve can be obtained without a space charge and then with a space charge the amount of charge in the oxide can be determined.

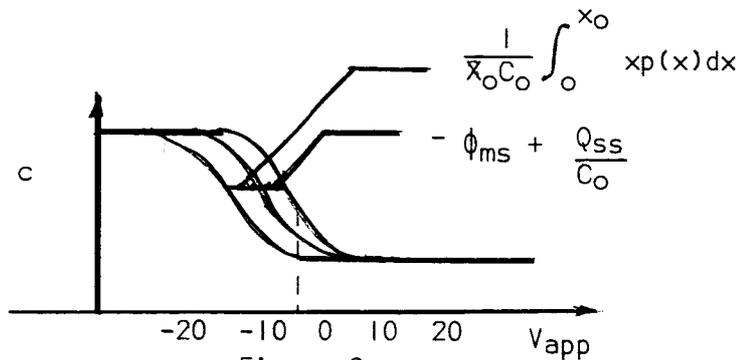


Figure 9  
High Frequency Capacitance Versus Voltage

In summary, ion migration is probably one of the major phenomena occurring in the M-O-S device. However, there are aspects of this phenomenon which are not well understood. For instance, in some cases the space charge in the oxide appears to be negative rather than positive. The charge could not be oxygen ion vacancies in this case of course. The space charge model does not replace the surface state model but supplements it. There are phenomena which can only be explained by the surface state explanation. The space charge phenomena go a long way in helping to understand the M-O-S structure but there must be further study before complete understanding is achieved.

## II. Process Furnace Arrangement

The connection unit was built for the purpose of removing impurities from the gases used with the furnace, to provide a means of mixing the gases, and to provide a means of introducing steam to the gases.

Since the primary reason for the unit is to purify the gases, care must be exercised so that no impurities are added by the connection unit itself. For this reason inert materials were used throughout the system. Stainless steel, teflon, and quartz are the only materials used. All of the pipes and fittings are of stainless steel, the stoppers and O-rings are made of teflon, and the furnace tube and all flasks are quartz. This system will provide little contamination of the gases and help to provide for reproducibility of results.

The gas to be used in the oxidation process is bought in cylinders. Each cylinder is connected to a flow meter which allows the flow rate to be controlled. The gas flows from the flow meter to the input unit.

The input unit provides for the selection of gas and the mixing of gases if required. At present only two gases can be used; however, the system can be expanded to provide for three and more gases with a minimum of effort.

From the input unit the gas flows into the cold trap. The cold trap consists of a quartz flask sitting in liquid nitrogen. The gas flows into the cold trap and is delayed by a pyrex wool bundle long enough for the water vapor and other impurities which have a boiling point higher than nitrogen to condense and remain in the flask.

The gas then leaves the cold trap and goes into the furnace. Between the cold trap and the furnace a steam source can introduce steam into the gas.

This is optional and is controlled by a valve. The steam source is a quartz flask filled with deionized water and sitting in a heater which controls the temperature of the water.

There are several glass to metal connections in the system and since the materials were restricted to stainless steel, teflon, and quartz, special clamps had to be designed and built for these connections. The connections occur at the entrance to the furnace tube, the cold trap, and the steam source.

The most difficult connection was at the entrance to the furnace tube. The quartz furnace tube has an outside diameter of approximately two and one-half inches. The connection had to be easy to connect and remove because the arrangement of the furnace in the lab is such that the specimens have to be entered from the same end as the gas and the thermocouple used to profile the furnace. Figure 10 presents the design. The connector slips over the end of the quartz furnace tube and is then tightened. This compresses the teflon O-ring which makes a pressure contact with the quartz tube. The connector is of stainless steel. The upper rod extending from the end of the connector is for the gas input and the lower rod is for the thermocouple.

The connectors for the two flasks are similar to each other. The difference is that two rods extend into the flask for the cold trap and only one for the steam source. The connector consists of a teflon stopper with a hole for the stainless steel tubing to enter. The aluminum neck on the flask allows tightening of the stopper. The arrangement is shown in Figure 10.

Optional

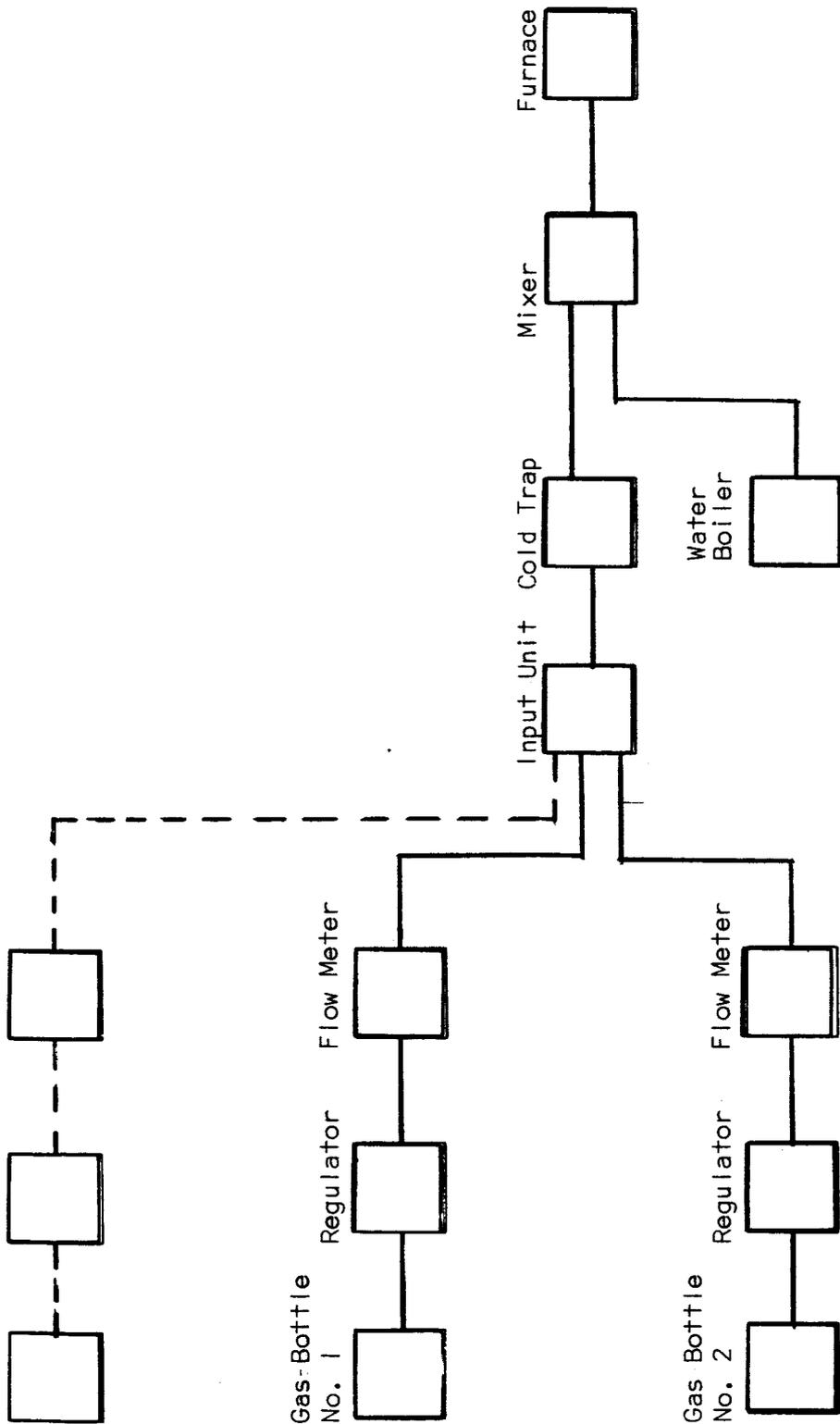
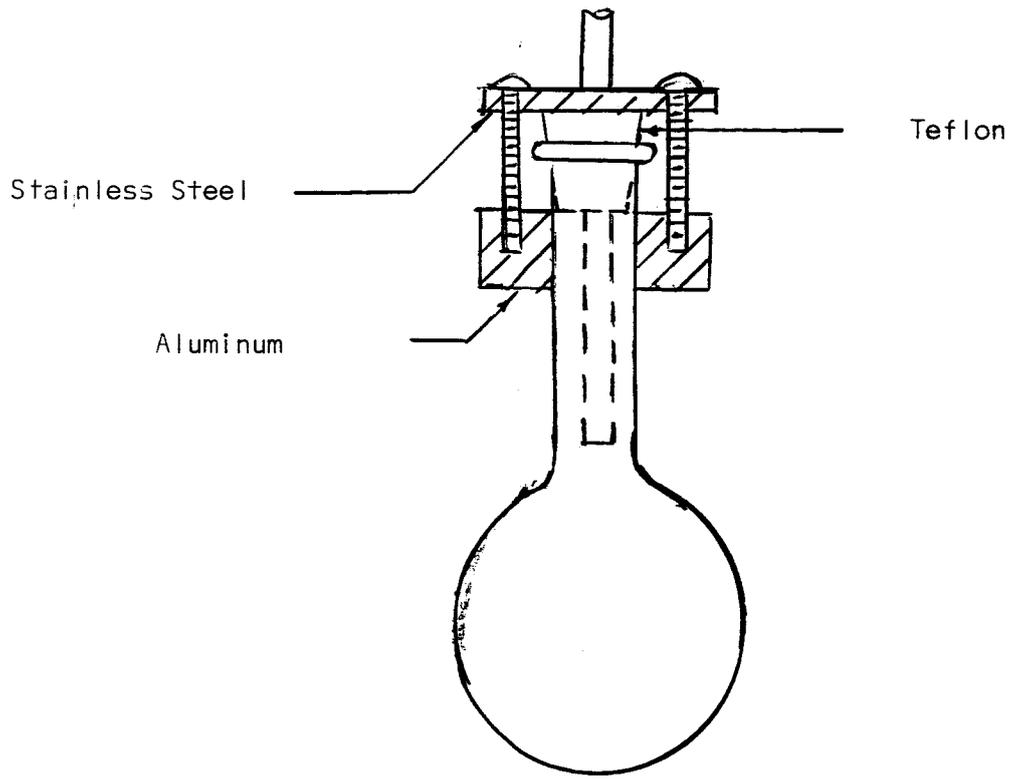
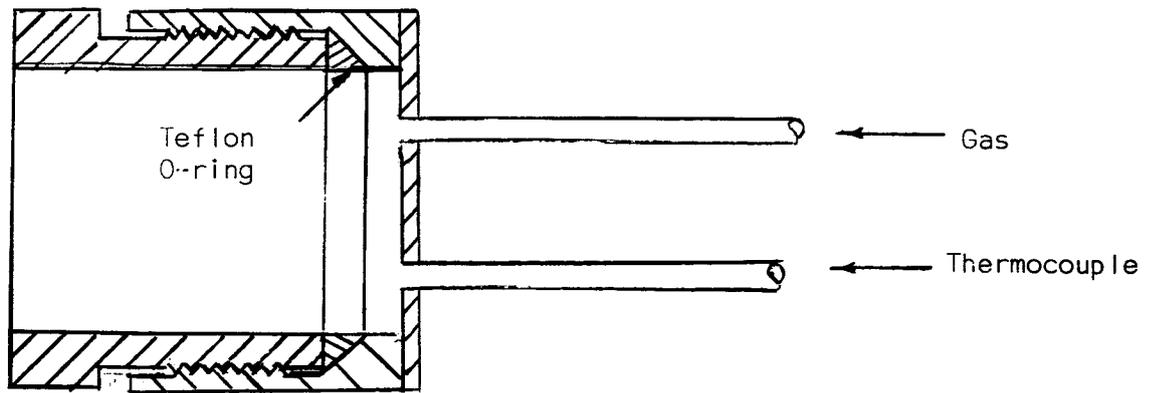


Figure 10a  
Process Furnace Arrangement



Flask Arrangement



End Piece for Furnace Tube

Figure 10b

### III. Capacitance Measurement

#### A. Introduction

Experiments as early as 1746 at the University of Leyden indicated that electrical charge could be stored in a device later called a Leyden jar.<sup>3</sup> This device is the same, in principle, as a modern capacitor. Today, capacitors come in many configurations yet they follow the same physical laws as the early Leyden jar.

Capacitance relates charge to applied voltage. The amount of charge ( $q$ ) stored in a capacitor is directly proportional to the potential difference ( $v$ ) applied to the terminals of the device. In equation form:

$$q = Cv \quad (16)$$

where  $C$ , the constant of proportionality, is called the capacitance of the device. This defining relation holds for all configurations of capacitors.

We restrict ourselves to devices which relate current and the first derivative of voltage by a positive, real time invariant constant as in equation (2).<sup>4</sup>

$$i = C \frac{dv}{dt} \quad (17)$$

where  $C$  is the constant described above. The capacitors under discussion are properly described this way.

For such a device, it is possible to measure the value of the constant  $C$  in equations 1 and 2 to several significant figures. The degree of accuracy of measurement is dependent on the method of measurement and the internal accuracy of the measurement device.

Measurements of this type require both accuracy and precision. Accuracy refers to how close a measurement has been to the true value of a

quantity. Precision refers to the clustering of measurements around an average. Thus it can be seen that a measurement can fall either close or further away from its true value for high and low accuracy and successive measurements of the same quantity can show a small or large deviation about their average for high and low precision. The ultimate goal for a measurement process would then be both high accuracy and high precision.<sup>5</sup>

There are, of course, other possible sources of error such as systematic and random errors; however, random errors occur in such a manner as to be readily detectable and easily compensated, and systematic errors are usually undetectable.

Examination follows of both the theory and practice of capacitive measurement and the problems involved. The design of the capacitance measurement system considers theory and present experimental results.

## B. Capacitance Measurement Considerations

### Capacitors

Most capacitors, when examined from a view external to the terminals, i. e., when considered as a black box, can be represented by the three capacitances shown in Figure 11. The direct capacitance  $C_{HL}$ , which is the actual capacitance of the unknown, and  $C_{LG}$  and  $C_{HG}$  the capacitance from the corresponding high and low terminals to the capacitor enclosure surrounding objects and ground.<sup>6</sup> This direct capacitance and the associate ground capacitances are the capacitive values constituting an unknown when the unknown is connected to a measuring device.

There are two modes of measurement for such an unknown: two terminal and three terminal. For two terminal measurement the C terminal is connected

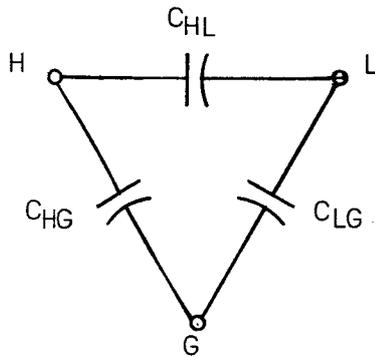


Figure 11  
Schematic Diagram of a Capacitor Showing  
Direct and Associated Capacitance

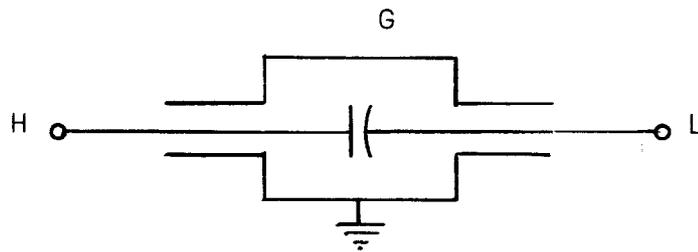


Figure 12  
A Three-Terminal Shielded Capacitor

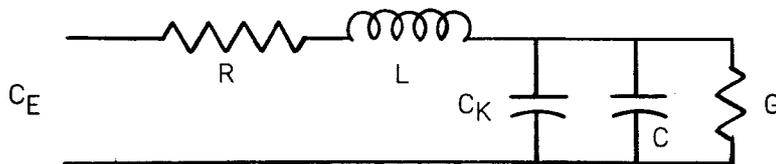


Figure 13  
Equivalent Circuit of a Two-Terminal Capacitor

to the case, shorting  $C_{LG}$  and the total capacitance is then  $C_{HL} + C_{HG}$ , where  $C_{HG}$  is the capacitance between the high terminal and surrounding objects, and can be changed by changes in the environment surrounding the objects. The most evident factor affecting the environment is the wire connections from the measurement device to the capacitor.

The effect of the surroundings and the resulting uncertainties in measurement they introduce can be eliminated by making three terminal shielded measurements. Figure 12 shows a three terminal capacitor with its associated shield. The capacitances  $C_{HG}$  and  $C_{LG}$  are eliminated from the measurement circuit by the use of guard circuits which balance the values of  $C_{HG}$  and  $C_{LG}$  and thus exclude them from the values of direct capacitance ( $C_{HL}$ ) measured. Thus, it is possible to measure  $C_{HL}$  for any capacitor by introducing the desired unknown into a shielded, grounded environment.

Although real capacitors sometimes approach the ideal, there are small deviations from ideal performance which must be examined. An equivalent circuit for a capacitor is shown in Figure 13, which shows the lumped constant two terminal form of a capacitor.  $R$  is the metallic resistance of the leads, supports and capacitor plates,  $L$  the series inductance of the leads and plates,  $C$  the actual capacitance of the plates,  $C_K$  the capacitance of the supporting structure, and  $G$  is the dielectric loss in the supporting insulator, air losses, and ac leakage conductance.

When a non-ideal capacitance is placed in a circuit, two external parameters are of concern. They are the effective value of capacitance at the external terminals and the losses inherent in the device. These quantities are the series reactance and resistance of the device. Figure

14a shows these schematically. One can also consider the device from a parallel equivalent standpoint. This representation is shown in Figure 14b.

Consideration of total impedance or admittance in the complex plane is shown in Figure 15. where

$$B = \frac{1}{X}, \quad G = \frac{1}{R}, \quad \text{and } Y = \frac{1}{Z}. \quad (18)$$

Also, the power factor for the capacitor is  $\cos \theta$  where  $\theta$  is the angle between the R and Z or G and Y phasors.

One can define the dissipation factor and storage factor of a capacitor as being ratios of resistance to reactance and susceptance to conductance respectively. In equation form:

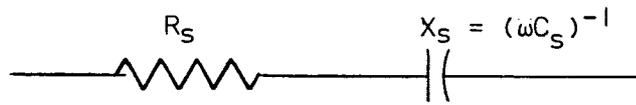
$$D \equiv \text{dissipation factor} = \frac{R}{X} = \frac{G}{B} = \frac{1}{Q} = \tan \delta \quad (19)$$

$$Q \equiv \text{storage factor} = \frac{X}{R} = \frac{B}{G} = \frac{1}{D} = \cot \delta \quad (20)$$

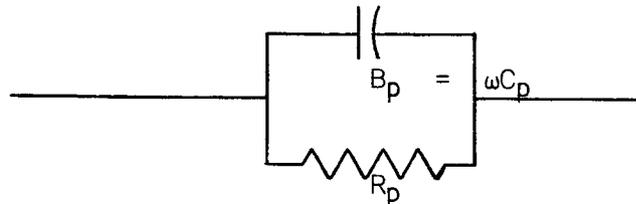
where equations (4) and (5) also give interrelations between D and Q.

Thus, it can be seen that the values that are actually measured by whatever system is being used are with respect to the terminals of the unknown.

Two representations for the device, series and parallel are common. The measured values of the unknown can be expressed in either set of parameters, depending only on which is more convenient. We can measure only the resistance and reactive components of the unknown. Figure 15A is the phasor diagram for the series case while the parallel equivalent is represented in Figure 15b.

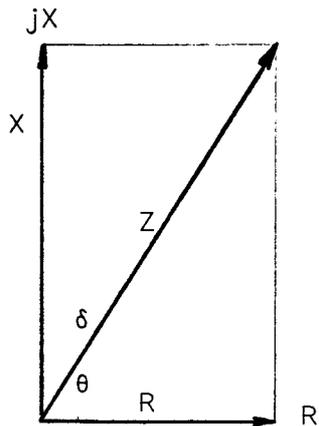


a. Series Representation

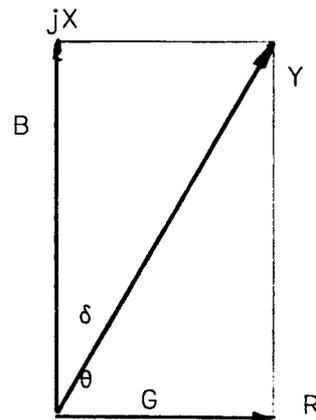


b. Parallel Representation

Figure 14  
External Terminal Representation of  
a Capacitor as a Circuit Device



a. Series



b. Parallel

Figure 15  
Imittance of a Capacitor in the Complex Plane

In the series case the dissipation factor D, the cotangent of the dielectric phase angle  $\theta$ , is

$$D = \cot \theta = \tan \delta = \frac{R_s}{\frac{1}{\omega C_s}} = \omega C_s R_s \quad (21)$$

where  $R_s$  and  $C_s$  are the measured values for the series equivalent representation. For the parallel case D is given by

$$D = \cot \theta = \tan \delta = \frac{G}{\omega C_p} = \frac{1}{\omega C_p R_p} \quad (22)$$

The interrelations between the series and parallel cases are given by:

$$C_p = \frac{C_s}{1 + D^2} = \frac{C_s}{1 + \tan^2 \delta} = C_s \cos^2 \delta \quad (23)$$

$$R_p = \frac{R_s(1 + D^2)}{D^2} = \frac{1 + D^2}{D \omega C_s} \quad (24)$$

Due to the series inductance, L, of the leads as shown in Figure 13, the terminal capacitance or effective capacitance of the unknown becomes greater than the zero frequency capacitance,  $C_0$ . When the frequency, f, is much less than the resonant frequency,  $f_0$ , [which is defined by  $\omega_0^2 LC_0 = 1$ ] the fractional increase in capacitance  $\Delta C$  is given by

$$\frac{\Delta C}{C_0} = \omega^2 LC_0 = \frac{f}{f_0}^2 \quad (25)$$

Similarly, the loss D is dependent on frequency, the dependence being:

$$D = \frac{G}{\omega C} + R_1 f \omega C \quad (26)$$

where  $R_1$  is Resistance at 1 megacycle and f is frequency in megacycles.

This describes a capacitive device as examined from a terminal point of view. When this type of device is connected to a measurement system,

the parameters measured are the effective capacitance, and the dissipation factor in either series or parallel configurations.

#### Measurement Devices

High accuracy and precision measurement of capacitive device parameters is generally made by the null method using a variation of the basic ratio bridge. The basic ratio bridge balances for a null or zero dc voltage across a detector by use of either a variable standard capacitor or a fixed standard capacitor and a variable ratio arm. The basic ratio bridge is shown in Figure 16.

After the insertion of  $C_x$ , the unknown, in the bridge circuit,  $C_n$ , the variable capacitor, is adjusted to obtain a minimum reading on the detector. When this null condition has been established, the bridge is balanced and the value of the capacitor  $C_n$  is the value of the unknown. This type of bridge is useable over a wide range of frequency ranges with an accuracy of about 0.01%.

For measurements of greater precision the usual approach is to use an inductively coupled transformer ratio arm bridge. With such a bridge, (Figure 17) accuracies in parts per million can be obtained for ratios as high as 1000:1.

In Figure 17, the primary winding of the transformer (usually toroidal to avoid flux leakage) is used only to excite the core. The voltage in the secondary arms of the bridge is then exactly proportional to the number of turns in the secondary windings. The ratio of the open circuit voltages is equal to the number of turns between taps, and this is easily and precisely determinable. When the number of turns between taps is fixed, the

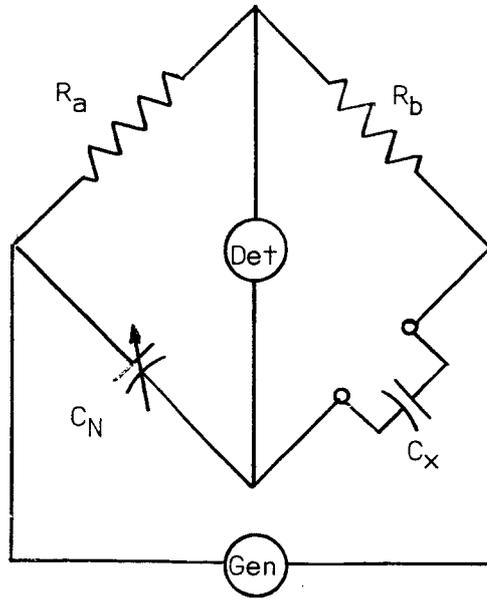


Figure 16  
Basic Ratio Bridge

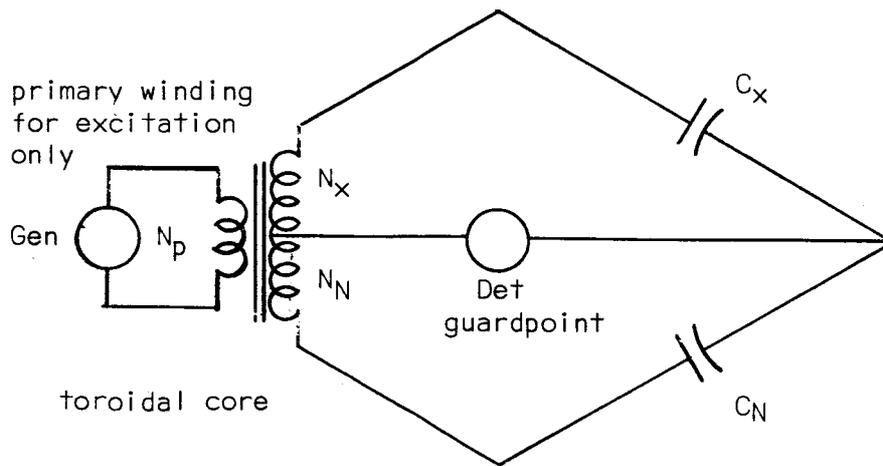


Figure 17  
A Simple Representation of a Transformer Ratio Arm Bridge

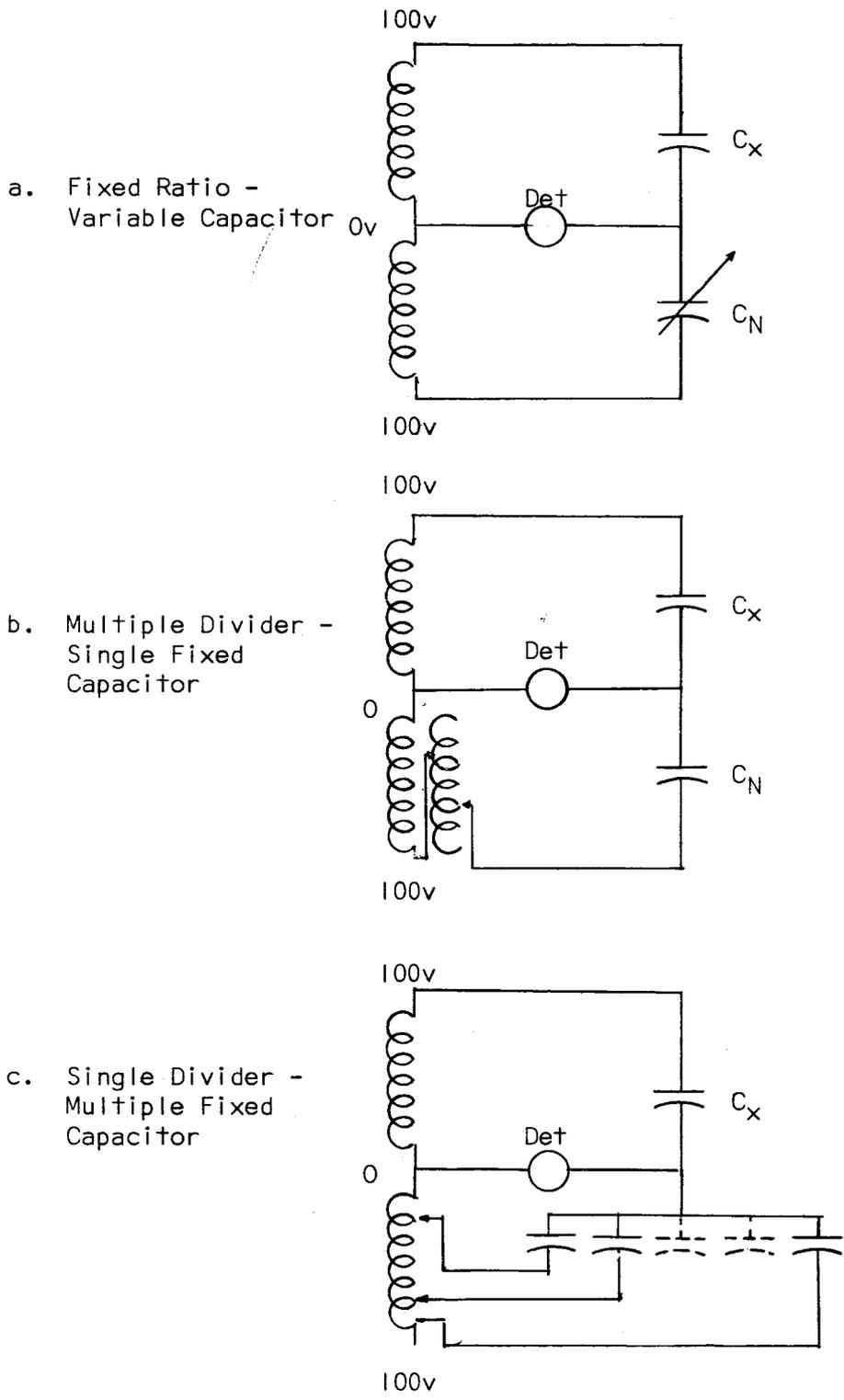


Figure 18  
Methods of Balancing a Transformer Arm Bridge

voltages appearing across these turns is invariant.

The condition for balance, or zero detector current, is then

$$V_N C_N = V_X C_X \text{ or } \frac{C_X}{C_N} = \frac{V_N}{V_X} = \frac{N_N}{N_X} \quad (12)$$

With this system, the internal capacitances from the guard point or center tap of the secondary are exactly balanced and do not contribute to the degree of error of the bridge. This bridge measures  $C_X$  in terms of  $C_N$  without an additional balance. Variations on this general procedure are shown in Figure 18. The three variations shown are:

1. Fixed ratio - variable capacitor
2. Multiple divider - single fixed capacitor
3. Single divider - multiple fixed capacitor

#### The General Radio Model 1615A Transformer Arm Ratio Bridge

Figure 19 shows a simplified circuit of the General Radio Model 1615A capacitance bridge. As can be seen from Figure 19, the General Radio Model 1615A uses a single decade of transformer voltage division and multiple fixed standard capacitors to provide six decades of resolution of capacitance. Through use of the variable resistances G and D four decades of loss or conductance are obtainable. There are eight standard capacitors, of which only six may be used at any one time to give six figure resolution for capacitance values. There are four precision decade resistors connected in series to give four significant figures for D and four precision decade resistors connected in parallel to give four significant figures for G.

On this bridge either  $C_p$  and G may be measured on  $C_s$  and D, depending on how "lossy" the device being measured happens to be. The choice is G for

more lossy devices.

Other features of the General Radio 1615A Bridge are:

1. Two or three terminal measurement.
2. Use of external standards to extend D and/or G ranges.
3. Use of external standards for measuring difference between the unknown and the external standard.
4. Wide frequency range.
5. Easily adopted to use with external power supply for dc bias of unknown being measured.
6. The General Radio 1615A Capacitance Bridge is furnished in the General Radio Model 1620A Capacitance Measurement System with the appropriate oscillator and detector, i. e., the system is furnished complete as the General Radio Model 1620A Capacitance Measurement System.

Comparison of the General Radio Model 1620A Capacitance Measurement System with Other Available Bridges and Reasons for Selection

On the basis of manufacturers' literature, three capacitance measurement bridges were chosen for more critical evaluation. These were: the General Radio Model 1615A, the Electro-scientific Industries Model 277, and the Wayne Kerr Model B221A. At first examination, all of these bridges appeared to meet the requirement for the project capacitance measurement system.

The Wayne Kerr Bridge is a transformer ratio arm bridge suitable for two, three, or four terminal measurement, and is capable of making measurements in any quadrant of the complex plane. This bridge makes use of two precision transformers - one for signal input and the other to the detector. The

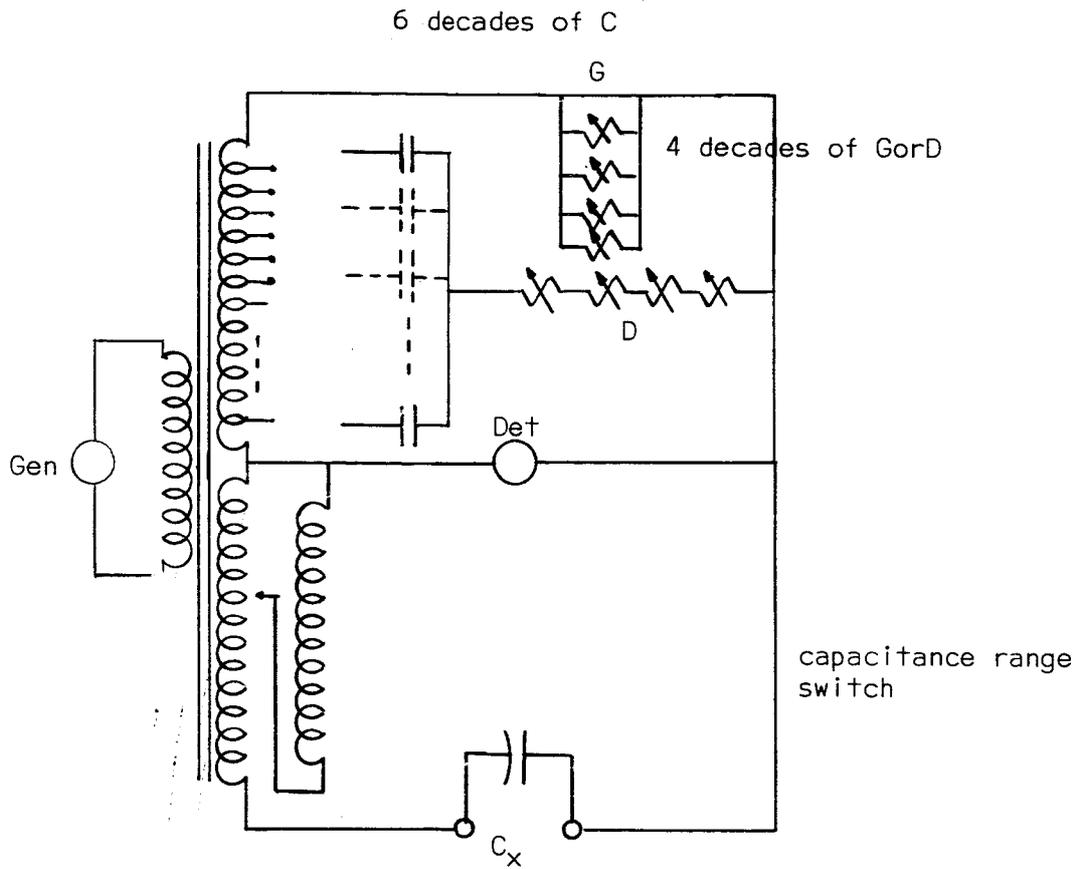


Figure 19  
Simplified Circuit Diagram of the General Radio  
Model 1615A Transformer Arm Ratio Bridge

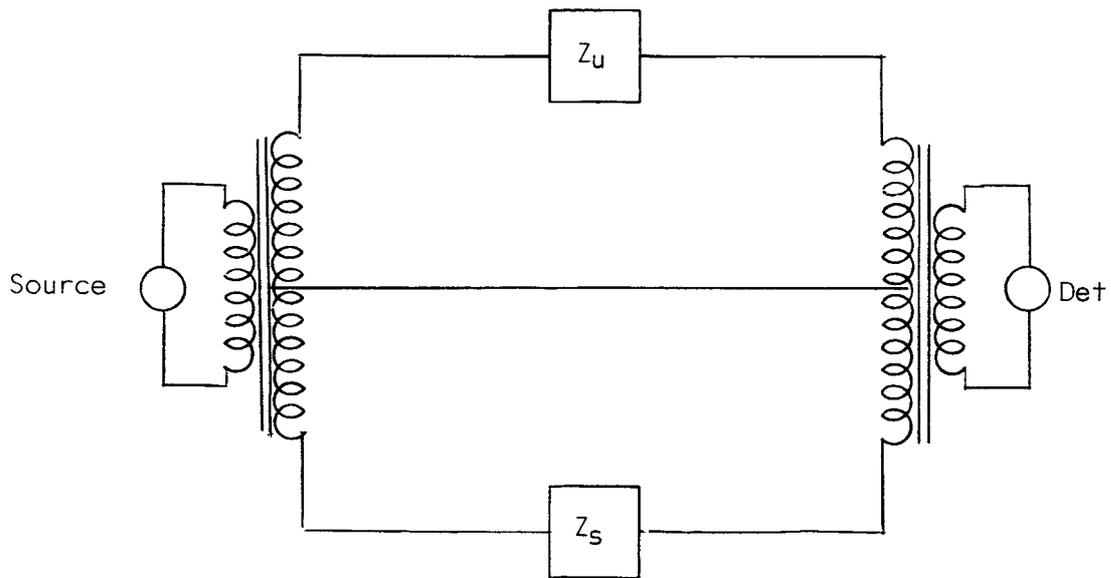


Figure 20  
Simplified Circuit Diagram of Wayne Kerr Model

bridge balances the unknown  $Z_U$  against a standard  $Z_S$ . The use of two transformers gives a comparison range of  $10^6$  to 1. (Figure 20)

The manufacturer specifies four significant figures with an accuracy of the order of 1% or greater. It was felt that this bridge would not produce measurements of the required accuracy, and it was not given further consideration.

The General Radio Model 1615A and the Electro-scientific Industries Model 277, with the recommended oscillators and detectors, were evaluated in the laboratory. The Electro-scientific Industries Model 277 bridge is a revision of the basic ratio-arm bridge as shown in Figure 21.

Unknown capacitance with its associated loss tangent is inserted in the unknown terminals and the bridge is balanced by a series of standard capacitors shown as  $C_{VAR}$  in Figure 21.

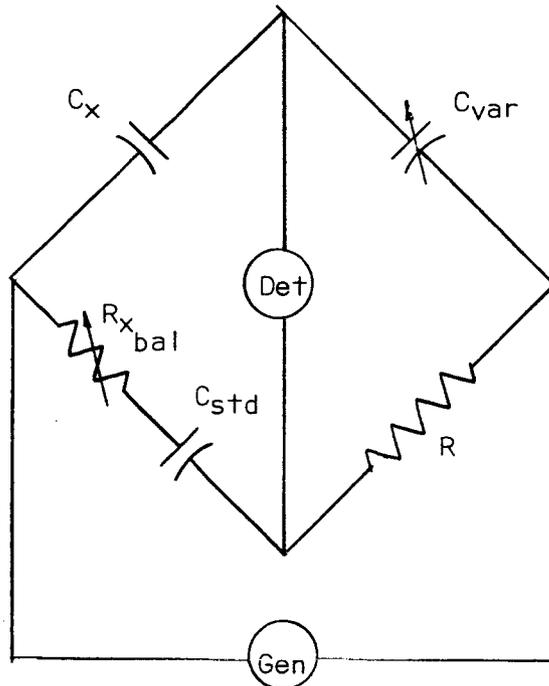


Figure 21  
Simplified Electro-scientific Industries  
Model 277 Capacitance Bridge

Data Produced by the Capacitance Measurement System

To check the operation of the bridge system, a p-n junction diode capable of withstanding a reverse bias of the order of several hundred volts was placed in the sample holder. A series of capacitance measurements were made. The log-log plot of the bias voltage applied in the reverse direction versus the value of capacitance measured is shown in Figure 25. As can be seen from the plot the slope is approximately  $-1/3$ . This implies that the p-n junction measured is of the graded junction variety. For a graded junction the expression for capacitance is:

$$C = \left[ \frac{q a \epsilon^2}{12 |v|} \right]^{1/3} = k \left[ |v| \right]^{-1/3}$$

After thorough comparison of these bridges, the General Radio Model 1615A Capacitance Bridge Measurement System was selected for use for the reasons listed below:

1. The General Radio Bridge allows measurement of capacitance to a greater number of significant figures.
2. The General Radio Bridge has a definite advantage over the Electro-scientific Industries model in regard to the method of application of bias, i. e., it requires very little in the way of an external bias circuit.
3. The General Radio Bridge has greater degree of accuracy since the accuracy in the Electro-scientific Industries bridge is in terms of  $\pm 1$  dial division, and the General Radio bridge is in terms of the value of the unknown.
4. The General Radio Bridge has a better system of three terminal measurement with the use of shielded cables that cancel the terminal to ground capacitances.
5. The measurement of D, the "dissipation factor" on the Electro-scientific Industries bridge is less accurate than on the General Radio bridge, i. e., it has a very broad null.
6. The digital readout of the General Radio bridge is much easier to read than the Electro-scientific Industries bridge.
7. The General Radio bridge system is more difficult to obtain an initial balance, but the successive "difference" measurements are easier.
8. Frequency selection is much greater in the General Radio system.

## Project Capacitance Measurement System

Figure 22 shows a block diagram of the system presently in use. The 1615A Capacitance Bridge is, of course, the central piece of equipment. The ac signal for measurement is supplied by a General Radio Model 1210C oscillator to cover a frequency range of 20 cps to 100 kc. The oscillator is powered by a General Radio Model 1203B unit power supply. For our purposes the output range is zero to seven volts with sinusoidal waveform.

The dc bias is supplied by Kepco 0-30 Model ABC power supply and a Kepco Model B 0-400 volt power supply for the higher voltage ranges. Both of these power supplies are continuously variable over their entire ranges.

The General Radio Model 1232A tuned amplifier and null detector is used with the bridge. After the maximum bridge setting below the noise range is obtained on the gain of this instrument, the output is monitored by a Hewlett-Packard 425A VTVM dc microvolt ammeter. For extremely sensitive balances this output is monitored by a Keithley Model 149 dc milli-microvoltmeter.

Due to the identical nature of the terminals for shielded measurement and unshielded three terminal measurement on action of the measurement switch, the ac and dc voltage levels appearing across the sample are monitored by Hewlett-Packard 400H VTVM and Hewlett-Packard 413A DCVMS respectively. The switch is then returned to the shielded position for measurement.

### C. Laboratory Measurements

It is a long step from the theory of capacitance measurement to the production of satisfactory experimental data. The problems encountered in attempting to produce data are of assorted variety and different natures.

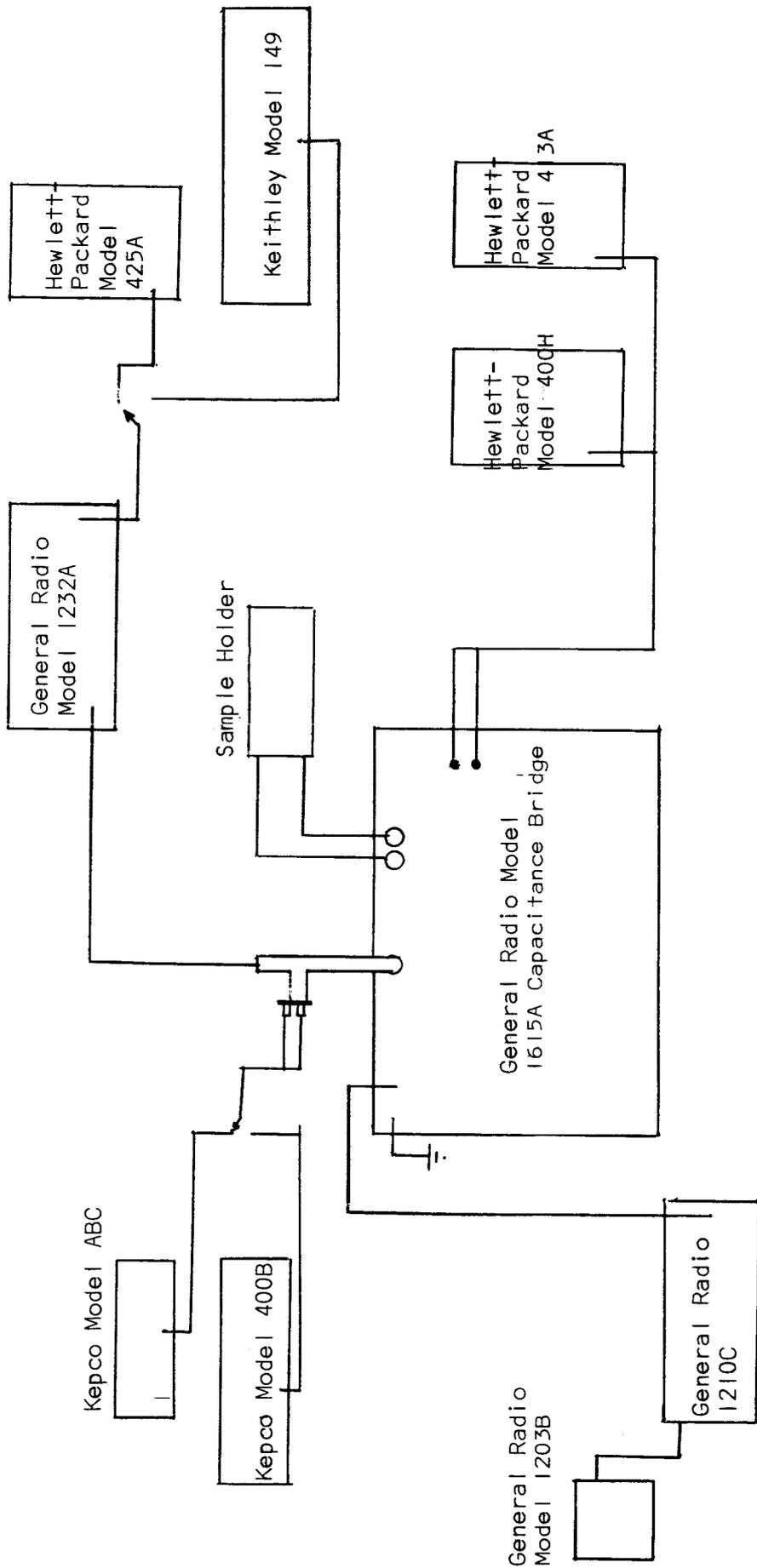


Figure 22  
Capacitance Measurement System

In state-of-the-art measurements such as are being attempted, some factors have to be considered that might ordinarily be ignored. Three terminal shielded measurement of capacitance was selected as mode of measurement for our project due to the isolation from surrounding stray signals produced by the shield around the device.

Some of the problems encountered during the process of setting up and checking out the capacitance measurement system are discussed below, and the present solutions to the problems are given.

#### Sixty Cycle Hum

One of the ever present problems of electronics measurement and design is the 60 cps ac voltage induced in circuits due to the presence of 117 volt 60 cps power provided as the normal power supply in all buildings. Although the magnitude of this induced voltage is not large, and can normally be neglected, due to the magnitude of the signal level we use in measurement (on the order of several millivolts), the magnitude of the 60 cps "hum" becomes significant. This problem was minimized by moving the entire capacitance measurement system to a screen room having 120 db attenuation at 400 mc and taking great care in the arrangement of component units of the system and in the grounding of the system. As a further precaution, all leads interconnecting the portions of the system specifically necessary for measurement have been shielded. Also, all ground loops have been avoided, as far as is known. The only 60 cps signal present in the screen rooms is the supply for the instruments. As this is a regulated supply there are almost no stray fields, since the input is 3 wire, single phase, 117 volts. The lights in the screen room are 12 volt dc to eliminate any other source

of ac voltages.

#### DC Bias

To provide the dc bias voltage for our samples, the dc power supply was connected as shown in Figure 22. Although the application of dc bias was simple enough, a method for the measurement of the actual value of dc bias existing across the sample was not readily apparent. It was found that the identical dc bias potential across the shielded terminals appears across the unshielded three terminal binding posts. Thus, a dc voltmeter may be connected across the three terminal binding posts and by use of the selector switch between types of measurement the existing dc bias may be measured without loading the voltage across the probes in the sample holder. This means that once the sample is inserted in the sample holder the conditions of bias existing across the sample may be determined by using the selector switch. DC bias will be measured to approximately four significant figures.

#### AC Signal Levels

It was thought at first that signal level could be set and measurements carried out for capacitance. Due to the nature of the bridge used, however, signal level must be varied to obtain the best balance. The procedure used will be to balance the bridge using variation of the signal level and then while attempting to hold the balance to minimize the applied ac signal level. It is believed at this time that the minimum signal level that can be obtained and consistently held will be of the order of several millivolts.

#### Circuit Parameters

As was seen in Figure 13, the equivalent circuit of a two terminal

physical capacitor has associated parameters to account for the non-ideal behavior of the device. At this point in the program and due to the comparative magnitudes of these parameters, these parameters  $R$ ,  $L$ , and  $C_K$  shall be neglected.  $R$ , the resistance of the leads, is of the magnitude of one ohm and is much less than the sample resistance.  $C_K$ , the capacitance of the mounting and all associated, is neglected due to the nature of shielded three terminal measurements. The capacitance of the leads and mounting is balanced out by appearing on both sides of the unknown. The inductance of the leads is neglected as a second order effect at this time.

#### Sample Holders

The use of three terminal measurements requires some type of sample holder and associated shield. The requirements of this device are simple. First, it must provide shielding from environmental electrostatic fields; second, it must provide a mounting for the sample; third, it must provide electrical connections for signal and bias. Figure 23 shows a schematic of the sample holder presently in use.

This is a relatively simple sample holder, and Figure 24 shows a tentative revised version that is in the process of being designed. The second version (Figure 24) is to be capable of being used at low temperatures.

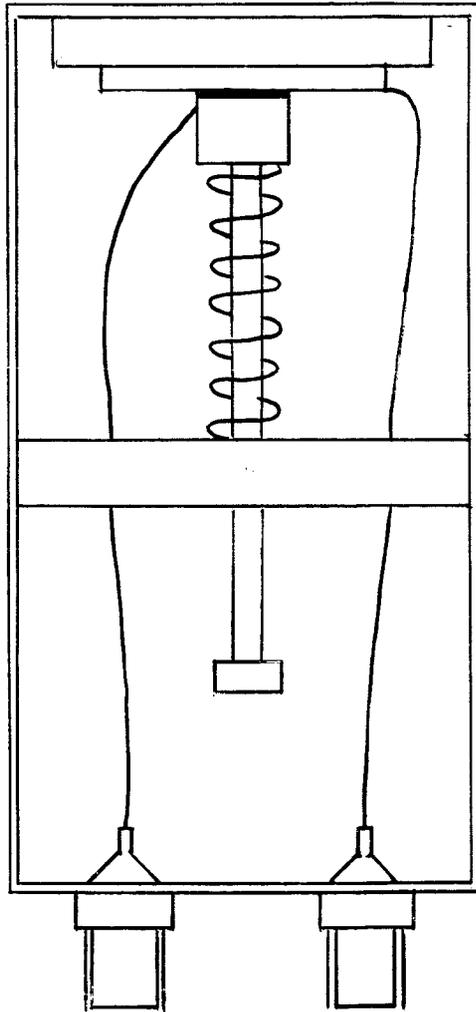


Figure 23  
Sampler Holder for M-0-6 Diodes

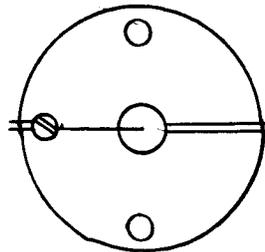
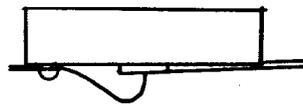
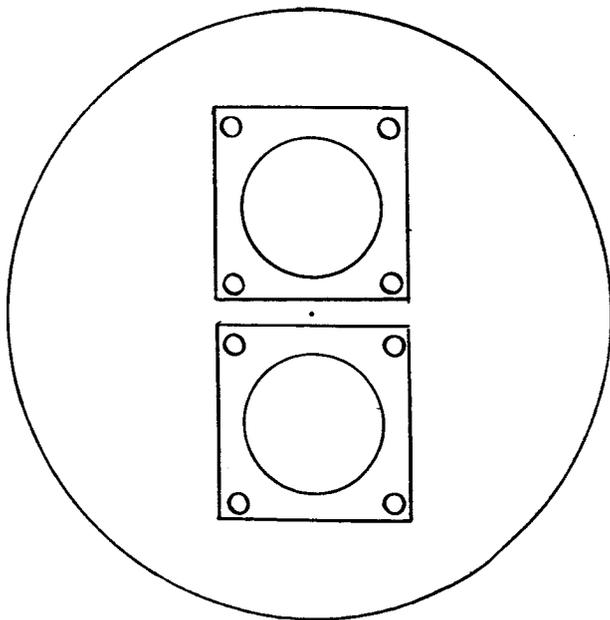
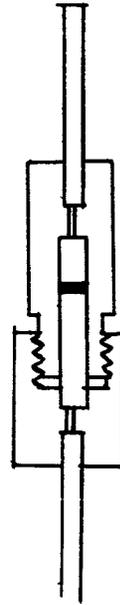
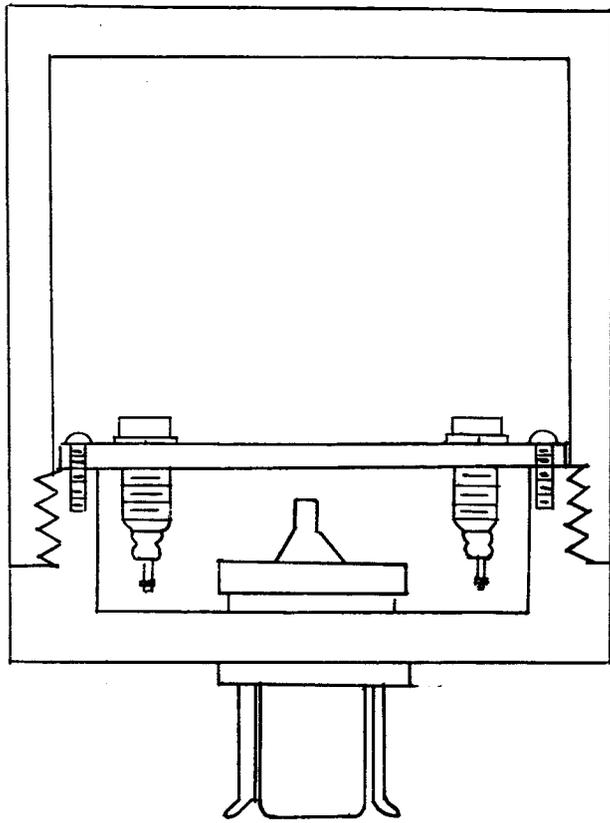


Figure 24  
Revised Sample Holder

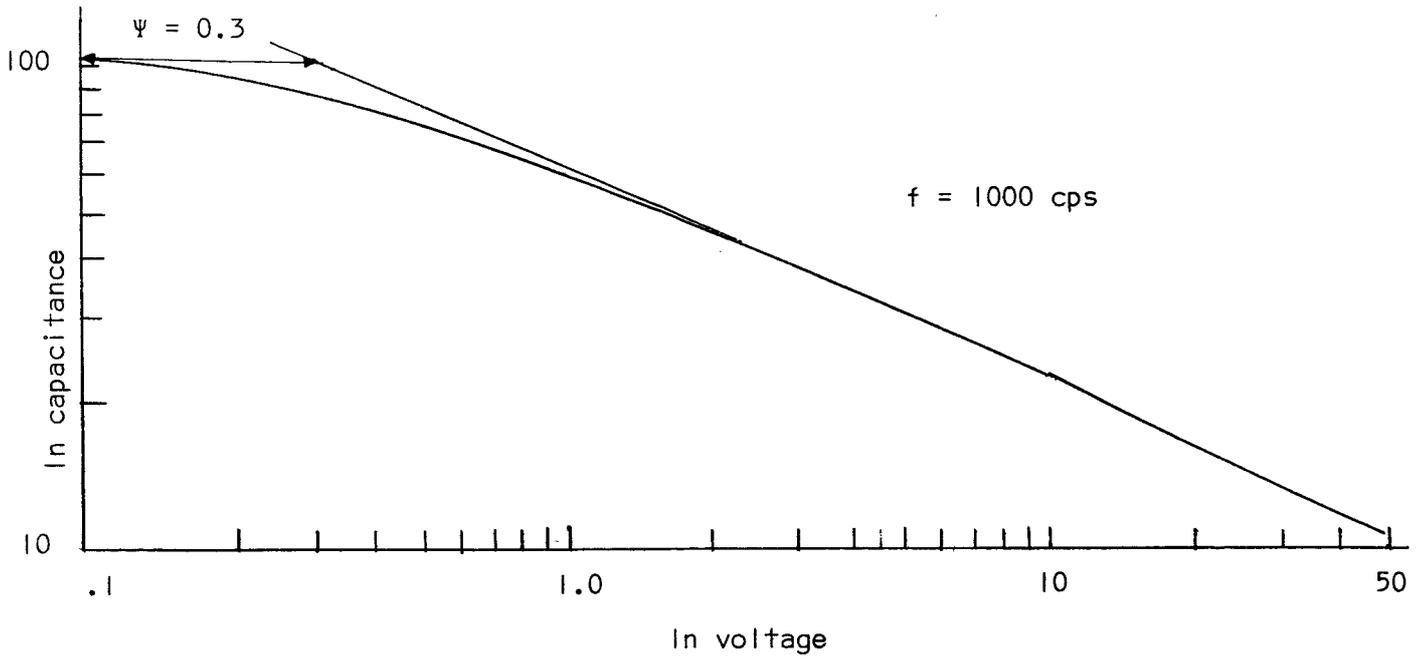
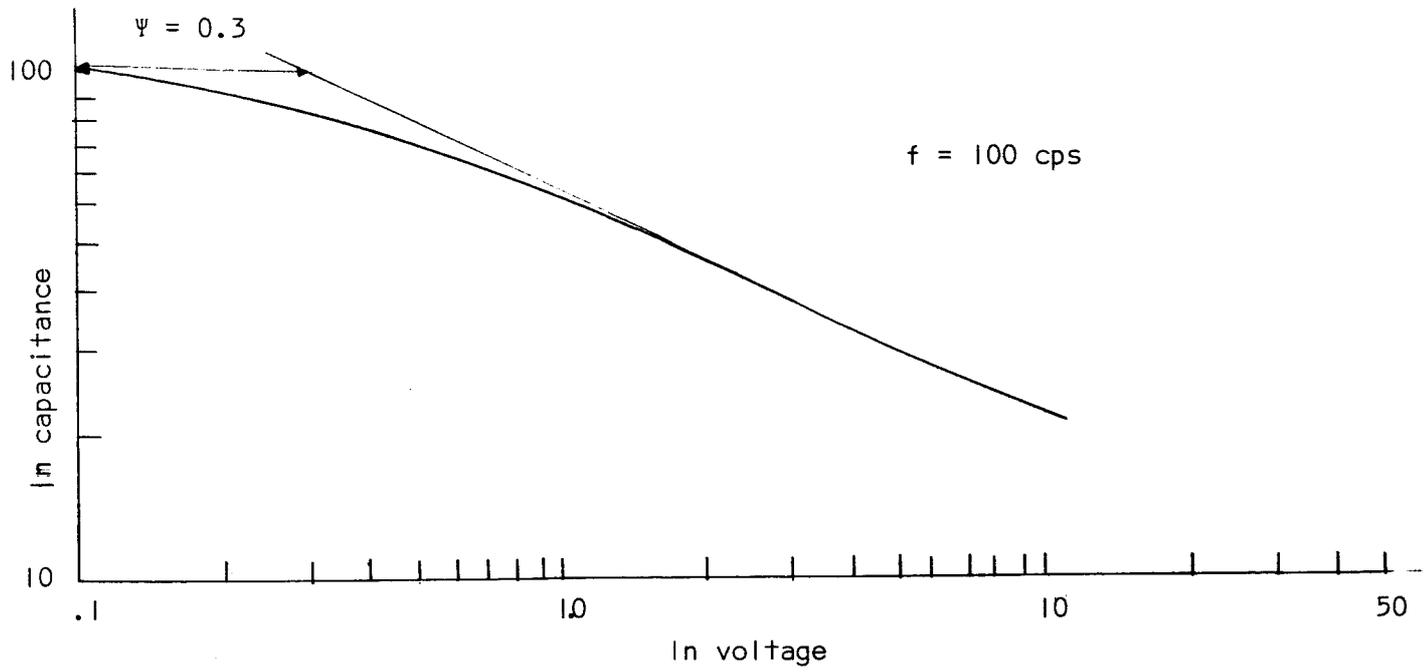


Figure 25a and 25b  
Capacitance Versus Voltage of a P-N Junction

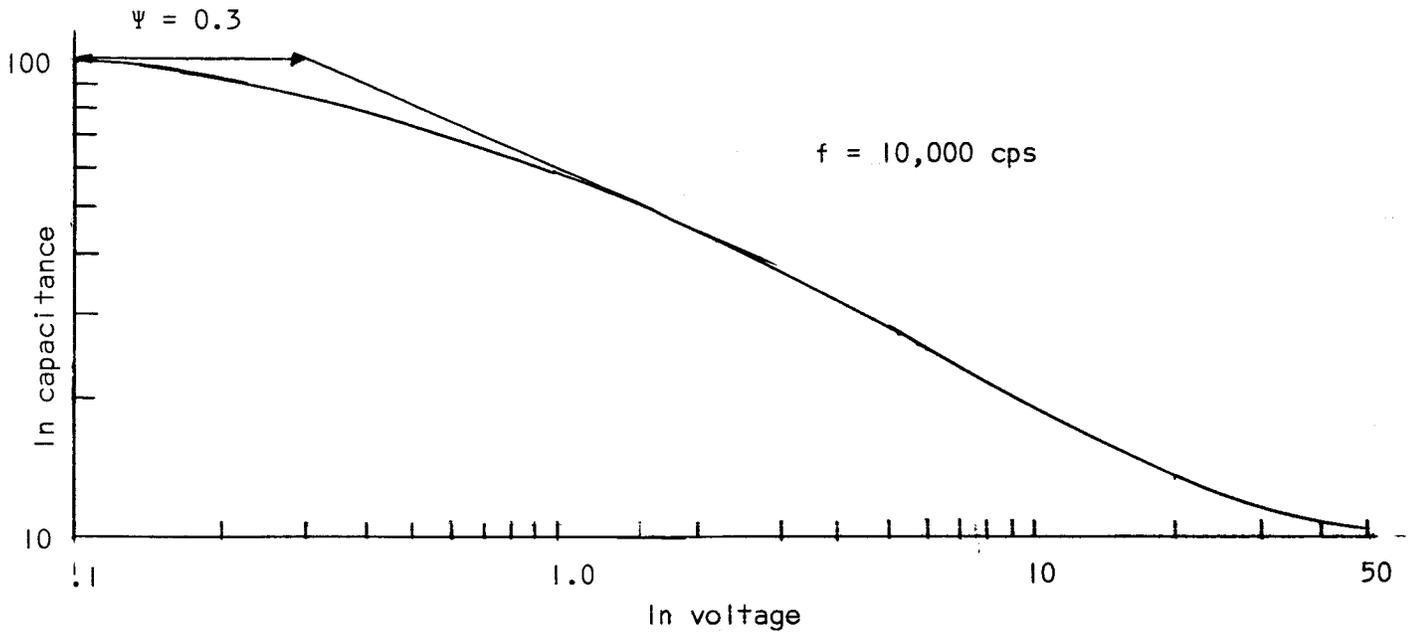


Figure 25c  
Capacitance Versus Voltage of a P-N Junction

Measurements were made at three frequencies: 100 cps, 1000 cps, and 10,000 cps.

It should be noted that as the values of capacitance measured become smaller the nulls become harder to obtain. At a frequency of 100 cps, it was not possible to obtain a null for bias voltages greater than 10 volts. These measurements indicate the need for the development of experimental techniques for the measurement of small values of capacitance.

Tables 1, 2, and 3 in the appendix give the specific values of voltage applied and capacitance measured.

#### Review and Discussion

In the preceding sections of this report we have discussed the nature of charge storage devices and the theory of equipment to measure the external parameters of these devices or capacitors. The advantages and disadvantages of several applicable capacitance measurement systems have been presented and the system selected for use has been described. Additional techniques of capacitance measurement for the production of accurate, useful data are being developed to supplement basic known techniques.

There are certain problems that need to be solved before complete confidence may be had in the capacitance-voltage data throughout the entire range of measurements. For values of capacitance, say around 1000 pf and greater, a good six significant figure value of capacitance may be obtained. However, as the value of capacitance becomes smaller the many significant figure balances become harder to obtain since a comparatively good null can be established for small capacitance with only three or at the most four significant figures. Thus, there is a problem of measuring devices with

high leakage current. The bridge does not balance sharply for capacitance values of the order of one hundred picofarads and less.

When measuring small values of capacitance, a null balance is difficult to obtain. After the first three significant figures are obtained, the balance, or null, condition is well established, and the production of successive significant figures calls for the utmost in sensitivity. An alternative method of producing data is possible with the General Radio Model 1615A bridge. It is possible to balance this bridge to read the difference between an external standard and the unknown. Judicious selection of the external standard would place the difference measurement in the range where a balance condition is reflected in five or six significant figures. Mathematical correction can then be made to produce experimental data on the unknown.

When the device producing the capacitance being measured has a dc leakage current caused by applied bias corresponding to values of loss tangent of the order of 0.40, the bridge cannot achieve a meaningful balance. This situation may be improved by measuring the parallel capacitance and conductance of the device, but conductance is limited to values less than 0.1 micromhos. If the loss of the device exceeds these values, a direct balance cannot be obtained. In these cases, the physical structure of the device might better be understood if data could be obtained and comparison made to a device with normal loss.

A possible method for obtaining this data would be to connect a capacitor with very low loss and large capacitance in parallel with the unknown. This would effectively increase the capacitance seen by the

bridge while decreasing the loss. This allows a balance to be obtained, and mathematical manipulation of data would give the capacitance and loss or conductance of the unknown.

These techniques, when fully developed, will expedite the generation of capacitance-voltage data. They will allow the examination of devices with wide latitude in parameters and contribute to the formation of the theory describing the physical processes that act in the device.

#### IV. Electron Microscopy

##### A. Etching Technique

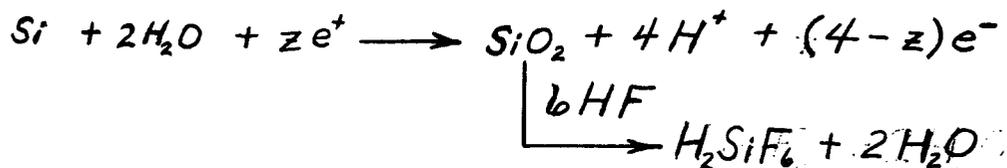
To complement the information obtained from the capacitance and voltage measurements, a simultaneous study of the interior structure of silicon is being made using the techniques of transmission electron microscopy. The primary problem in this study at this time is producing samples which are thin enough to transmit an electron beam. To yield useful information, the transmitted beam must have enough intensity to produce a recognizable image. Sufficient intensity is usually obtained if the samples are less than a 1000 Å thick; ideally the sample should be less than 500 Å thick. The electron microscope to be used in this study and the general procedure and apparatus which will be used to produce samples suitable for use in the electron microscope were discussed in the last report. In this section of this report a brief summary of the theory of electrolytic etching of silicon will be presented first. Next, the results of our efforts to date and the problems encountered will be presented.

##### Electrolytic Etching of Silicon

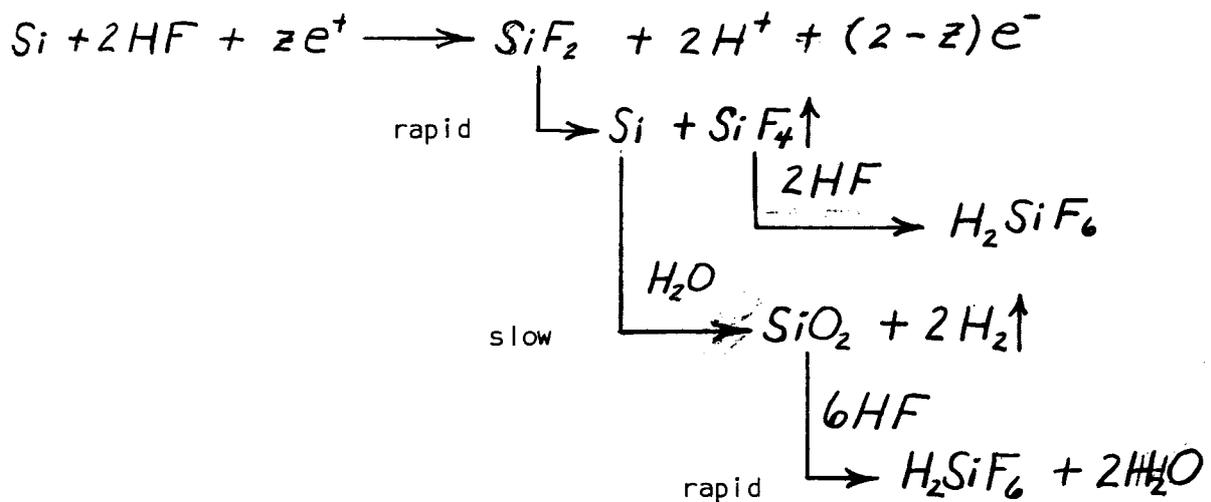
Electrolytic etching or electropolishing is often defined as the uniform removal of surface atoms with an atomically smooth surface as the desired end result. The most successful work reported in the literature has been accomplished using aqueous solutions of hydrofluoric acid, HF. Some work using strong alkaline solutions has been reported but these solutions chemically attack the silicon resulting in preferential etching and rough surfaces. Some fluoride salts are satisfactory for etching; however, the salts used and the final product of the anode reaction when

using these salts must be soluble in water. HF and its anode product hydrofluosillicic acid,  $H_2SiF_6$ , are highly soluble in water and it is for this reason that aqueous solutions of HF have been preferred for electrolytic etching of silicon.

It has been proposed that electropolishing of silicon begins when the HF begins to be used as fast as it arrives at the surface being etched. When this condition is met, a thin film, which is suggested to be  $SiO_2$ , is present on the surface. The exact dissolution process that occurs at the silicon surface is not known; however, the following process has been proposed. Each atom of a single crystal of silicon is covalently bonded to two underlying silicon atoms. When these bonds are broken by the arrival of a hole for each two-electron bond, the silicon atom combines with available oxide or hydroxide ions. The process may be as follows:



When the HF is not consumed as fast as it arrives at the silicon surface, a more complex reaction occurs resulting in the formation of a thick film on the surface. Films grown in electrolytes with a high concentration of HF ( $\sim 50\%$  HF) have been found to be almost all silicon. The following chemical process has been proposed.



The slow reaction of the Si and water is the reason for the build up of the silicon film on the anode surface.<sup>9</sup>

As mentioned above, holes are essential to the dissolution process. In p-type silicon holes are the majority carriers and consequently they are readily available. In n-type silicon, however, holes are the minority carriers and because of their short supply, some method must be used to generate and make them available at the silicon electrolyte interface if the etching process is to proceed at a reasonable rate. The easiest way to produce these holes is by high-intensity illumination of the working surface with light of wavelengths shorter than 1.2 microns. Those wavelengths longer than 1.2 microns only add heat to the sample and should be filtered out if possible.

#### Experimental Results

To date we have been able to produce several samples in which small areas were thin enough to be used in the electron microscope. Starting with

a silicon die 1/8" in diameter and about 0.030 inches thick, we can obtain a semicircular area with a thickness of less than 500 Å thick. The total usable area in these samples has been about one-tenth of a square millimeter. We seek to obtain an area one millimeter in diameter with a uniform thickness of approximately 500 Å. The major part of our efforts during the period covered by this report have been directed toward achieving this result. The problems encountered can be separated into three interrelated areas: (1) jet electrode position characteristics, (2) electrolyte composition, and (3) etching characteristics of the bulk and thin sections of the sample. This separation is arbitrary and only for the purpose of discussion. The discussion of the problems begins with a brief summary of the procedure used to obtain the thin samples.

Jet Electrode Position Characteristics: The current density, which will be discussed later under etching characteristics, is highly dependent upon the position of the jet electrode in relation to the working surface of the sample. Increasing the distance between the jet and surface by one millimeter can sometimes reduce the current by a factor of two or more. The electrolyte composition determines the magnitude of current change with distance of the electrode. It is desirable to have a high current density; therefore, in the samples produced to date the jet has been positioned about one mm from the surface of the sample as the desired thickness is approached. Having the jet this close to the surface introduces another problem, however. In this position the current density is not uniform throughout the electrolyte. The resistance of the electrolyte is such that the portion of the sample closest to the jet is subjected to

a higher current density. Because of this a circular ring on the sample surface is etched at a higher rate, thereby producing a thin section of non-uniform thickness. Although the thin section is small and non-uniform, the current density in this area is apparently uniform because preferential etching is not apparent when the sample is observed in the electron microscope.

**Electrolyte Composition:** The electrolyte used so far has been composed of Isopropyl alcohol, water and hydrofluoric acid. The composition of the electrolyte is critical for two primary reasons. With high concentrations of HF, a brown deposit is formed on the surface of the sample. The amount of deposit is a function of both the current density and HF concentration. For a given amount of deposit, the required current density increases as the HF concentration decreases. It is felt that this deposit is the same as that described during the discussion on the theory of etching. When the HF concentration is too low, non-uniform etching occurs. The sample will have a polished surface, but when the sample is examined with an optical microscope small dimples or impressions can be seen. In addition to the uneven surface, a considerable amount of time is required for the etching process.

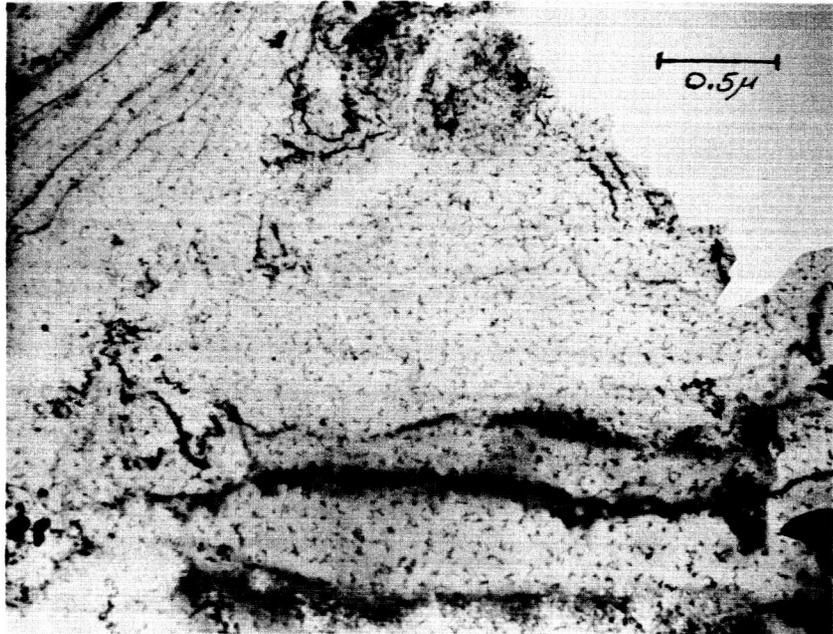
**Etch Characteristics of Bulk and Thin Sections:** In general, it appears that the surface condition is essentially a function of the current density for a given electrolyte composition. At low current densities small pits appear on the etched surface of the sample. Usually, these pits are large enough to be seen clearly with a 30 power optical microscope. However, at times they are so small and closely spaced that the sample surface appears to have been sand blasted, even when observed with a microscope.

As the current is increased these pits disappear and the etched surface takes on a mirror-like finish. When the current is increased still further, the brown film previously discussed appears on the surface and the amount of film increases with the current.

The conditions described above apply to etching of the bulk sample where the resistance of the sample is uniform over the surface being etched. If the surface being etched is thin ( $< 5000 \text{ \AA}$ ) the condition of uniform resistance of the thin section no longer holds. The resistance of the thin section increases as the thickness decreases. As a result, the current density in the thin section, which is the area of primary interest, decreases leading to preferential etching and the formation of etch pits. It is felt that these etch pits can be eliminated or at least reduced if additional fluoride ions are available from a modified solution composition.

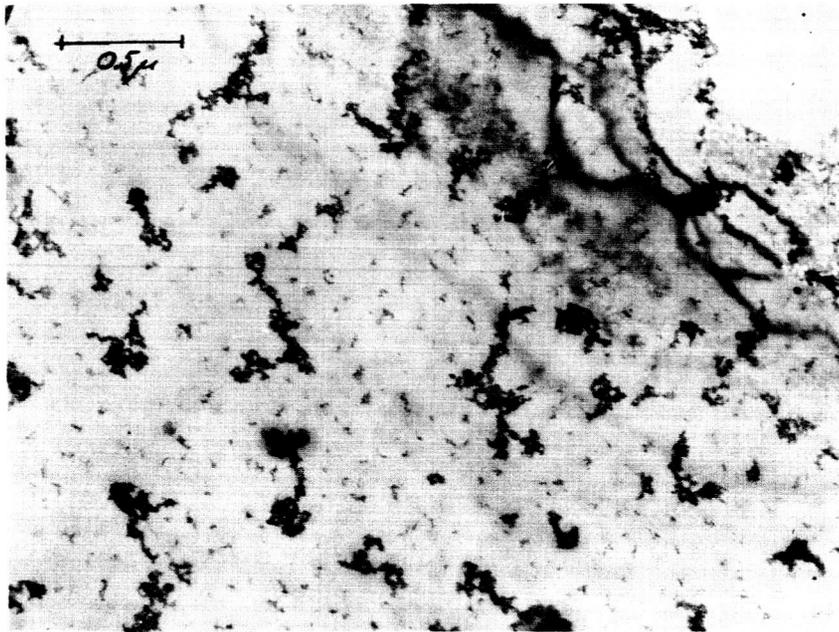
Summary of the Results: The best samples that have been produced so far have been etched using an electrolyte volume composition of 85 parts alcohol and 15 parts 48% HF. The current used has been about 40 ma with the jet electrode about 1 to 2 mm from the working surface. As mentioned earlier, the samples produced have had very small areas which are useable in the electron microscope. As an indication of our efforts to date, some photographs are presented on the following page. A complete interpretation of what can be seen in these photos is not available at this time.

Both photographs are from the same sample but different areas. There are three points of particular interest. The broad dark lines appear to be lines of constant sample thickness as they appear to move when the angle of incidence of the electron beam is changed. The narrow lines in the top



Thin Section of 30 ohm-cm., p-type Silicon (32,000X)

Figure 26a



Thin Section of 30 ohm-cm., p-type Silicon (32,000X)

Figure 26b

portion of the top photograph do not move when the angle of incidence is changed but instead they go in and out of contrast and for this reason they are assumed to be dislocations. The spots that appear in both photographs are believed to be segregated impurities. From the point of view of etching results, the area covered by each photograph is comparatively uniform in thickness. We have demonstrated how to produce thin samples suitable for transmission electron microscopy. The problem, as it stands now, is producing samples with larger areas of thin sections consistently and interpreting the structure observed in the electron microscope.

## B. Replication Techniques

A replica is a cast of a specimen which is opaque to electrons and permits their surface structure to be studied in the electron microscope. The replica consists of a thin film of electron-transparent material corresponding exactly to the surface topography of the specimen. By using this film one will be able to study irregularities in the silicon surface, the SiO<sub>2</sub> layer, and the Si-SiO<sub>2</sub> interface. A discussion of the Formvar/carbon replicating process was given in the August 31, 1964 Quarterly Report and will not be repeated here. Instead, other methods which have been tried will be discussed concerning problems encountered and results obtained. All of the methods tried have been two-step processes of one form or another since the single-step process (applying the carbon film directly to the surface of interest and then stripping or separating the film from the surface of interest) has the inherent disadvantage that the surface being investigated must be dissolved away and destroyed during the stripping process. The two-step process involves making a preliminary impression on the specimen surface in one material, coating the structure surface of this impression with the final replicating material, separating the two, and examining the replica in the electron microscope. This preserves the specimen for further use. The replicas made in the laboratory so far have been replicas of a thin slice of silicon cut on the diamond saw.

### Replicating Tape Method

A replica can be made by using a type of plastic "Replication Tape" sold by Ladd Research Industries, Inc. Either side of the "Replication Tape" may be used. The replica is made by placing a drop of acetone on the surface

to be replicated then immediately placing a small square of plastic on the acetone. The acetone softens the plastic which then follows the contour of the specimen surface thus making a faithful replica of the surface. Little or no pressure is necessary at this stage. The replica is ready to be stripped from the specimen surface when the plastic is dry. This usually takes about five minutes. The plastic replica formed is mounted, replica side up, on a glass microscope slide. "Scotch Tape" can be used to hold the plastic replica in place. A coat of carbon is then vaporized onto the replica side of the plastic tape. The actual carbon evaporation will be discussed later. At this point one of two things may be done.

(1) The route taken in this laboratory is as follows: At this point shadow the replica with a coat of platinum or platinum-carbon. Shadow casting is a technique which will be discussed later. The plastic-carbon replica is now removed from the vacuum chamber, and the plastic is ready to be dissolved away. This is done by placing the carbon-plastic replica in a small petri dish partially filled with acetone. The acetone dissolves the "Replicating Tape" and the carbon film floats free. The problem encountered so far has been the fact that the carbon film tends to come to pieces and breaks apart as it floats off of the tape. A piece of the carbon film large enough to study in the electron microscope can usually be "caught" on a microscope grid, but it is feared that later when interest lies only in a very specific area of the specimen that difficulty might be encountered in "catching" the right part of the film. The carbon replica obtained in this manner showed a good replica in the electron microscope. This is shown in the accompanying photograph.

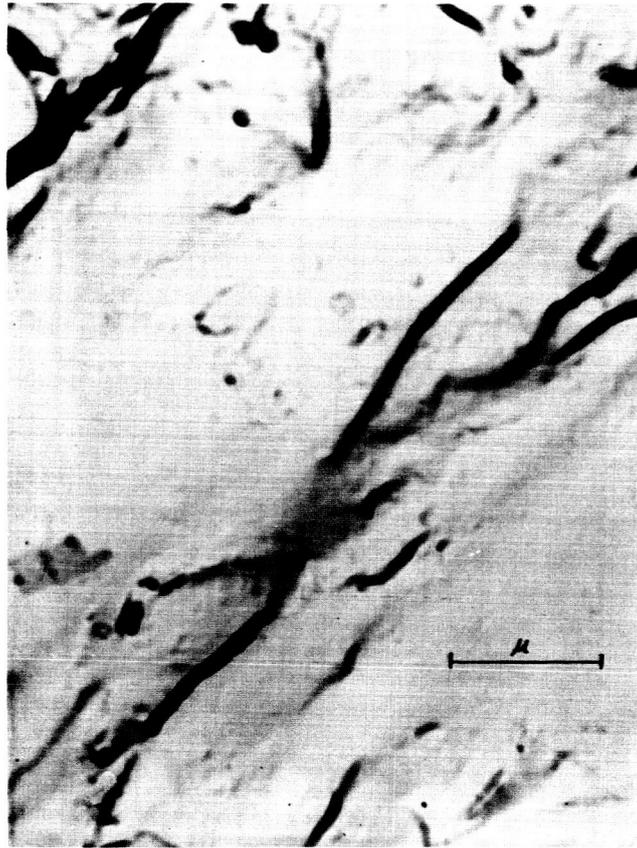


Figure 27

Replica of Silicon Sliced on Diamond Saw

Carbon Replica - Platinum Shadowing

Negative 10,000, Photo Enlargement 1.5X

The replica in Figure 27 was made in the manner previously described. The dark lines running diagonally across the photograph are  $3\mu$  apart. From the rotary speed of the diamond saw blade and the transverse speed of the blade, these lines can be justified as saw marks since one point on the edge of the blade cuts the specimen every time the specimen moves approximately three microns. The dark places are deposits of platinum from the shadowing process. Thus if one looked at a cross sectional view of the specimen from the direction of the arrow, a topography similar to Figure 28 would be observed. It must be kept in mind that this method produces a negative of the original surface.

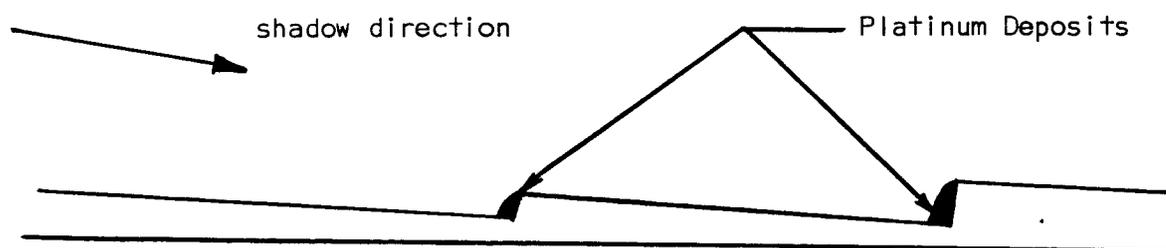


Figure 28  
The Silicon Surface

The other irregularities in the photograph are simply surface irregularities and imperfections on the silicon surface.

(2) An alternative route which may be taken after the coat of carbon is vaporized onto the tape is as follows: Remove the plastic-carbon replica from the vacuum chamber and then spread a thin (about 2mm) film of dental wax over the carbon side of the replica. This can be done by melting the wax in a small beaker and then spreading the wax over the replica with a warm glass rod. Dissolve the plastic part of the replica from the carbon-

wax part of the replica by placing the complete replica (plastic-carbon-wax) into a small petri dish partially filled with acetone. For ease of manipulation, place the wax side down so that it rests against the bottom of the petri dish. Use several changes of acetone over a period of several hours. All debris must be removed before proceeding to the next step. Now mount the carbon-wax replica on a glass microscope slide using Scotch Tape as described previously. The carbon side must face up and care should be taken so that the wax side of the replica does not stick very tight to the scotch tape. If the wax sticks too tight to the tape, it will be difficult to remove it from the tape without damaging the replica. Place the mounted carbon-wax replica into the vacuum coating unit and shadow lightly with platinum-carbon. This will be discussed later under "Shadow Casting." Remove the shadowed carbon-wax replica from the vacuum chamber and scribe the carbon part of the replica into small squares about 1/8" square. Alternatively, the complete carbon-wax replica can be cut into small squares. Place the scribed or cut replicas in a small petri dish containing xylene. The xylene will dissolve the wax and the carbon part of the replica will float free in the xylene. Transfer the carbon replicas through several changes of xylene or until the carbon replicas are clean. Slight heat will accelerate this procedure considerably. Now pick up the clean carbon replicas from the xylene solution with clean copper specimen grids and drain off excess xylene with filter paper. If the replicas are not yet thoroughly clean, then try an additional rinse with ethylene dichloride. This method has the advantage that perhaps the carbon would hold together better with the aid of the wax and also this method produces a replica of exactly the same shape as the specimen surface

whereas the first alternative produced a negative of the original surface (hills appear as valleys, valleys appear as hills).

#### Collodion Method

Another method of interest is the collodion method. This method is similar to the Formvar/carbon method described in the August 31, 1964 Quarterly Report, but it has enough differences to merit its discussion here. This method was first tried in the laboratory as follows: Place a drop of collodion on the sample, drain off the excess and let dry for about five minutes. Now place a piece of Scotch Tape over the collodion, press down, and carefully pull off the tape being sure that the collodion strips off with the tape. Now place the collodion replica and Scotch Tape in the evaporator and evaporate carbon onto the collodion. Next shadow with platinum or platinum-carbon. Take the tape-collodion-carbon out of the vacuum and dissolve off the Scotch Tape with ethylene dichloride. The collodion-carbon replica is then placed in a small petri dish partially filled with amyl acetate to dissolve the collodion and allow the replica to float free. The same difficulty was encountered here that was encountered with the "Replicating Tape" method, that is, the carbon seems to split apart as it floats free and, although it is easy to catch some piece of carbon, it is difficult to catch a particular piece. However, pieces of carbon have been caught on grids from this method and proved to be good replicas for study in the electron microscope.

In seeking to overcome this difficulty, the following method was used: Put a drop of collodion on the sample and let dry. Place a microscope grid on the collodion which is on the specimen. Put another drop of collodion

on the grid to hold the grid in place. Now place the Scotch Tape on the combination, carefully strip off, and proceed as before. It was felt that the grid being under the carbon would help contain the carbon and prevent it from shredding apart quite so much. However, this was not the case and nearly as much difficulty was encountered as before.

The following alteration was devised to contain the carbon replica on the electron microscope grid. The above method is used to the point where one is left with carbon-collodion-grid-collodion-Scotch Tape. At this point the tape is dissolved off with ethylene dichloride and one is left with carbon-collodion-grid-collodion. This combination is then placed in a jig as shown in Figure 29. By heating the amyl acetate and passing vapors over the replica the collodion will be dissolved away and one will be left with the carbon replica on the grid ready for examination in the electron microscope.

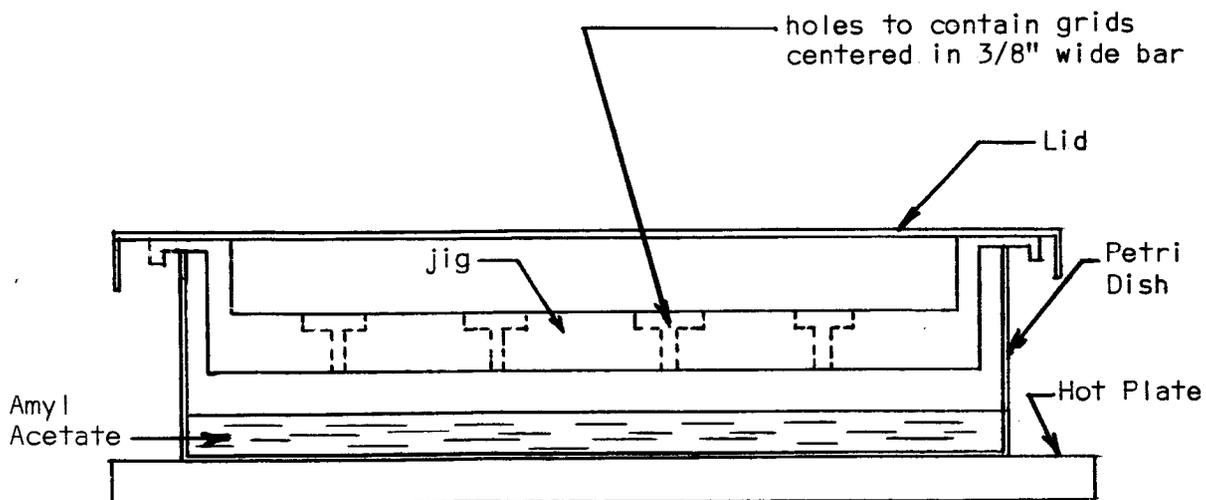


Figure 29  
Washing Apparatus

The difficulty encountered with this method is that the Scotch Tape will not dissolve away from the grid readily and hence one is left with collodion and tape on the grid which becomes "trash" or "dirt" when examined in the electron microscope. This can be overcome by cutting a very small hole in the tape (smaller than the size of the grid) and placing this small hole over the center of the grid when applying the tape.

Some difficulty is also encountered in stripping the collodion off the silicon surface. This may be overcome by breathing heavily on the collodion surface until a slight moisture condensate forms. Then apply the tape and proceed with the stripping process.

#### Vacuum Evaporation of Carbon

The apparatus used for the vacuum evaporation of carbon is shown in Figure 30 in simplified form.

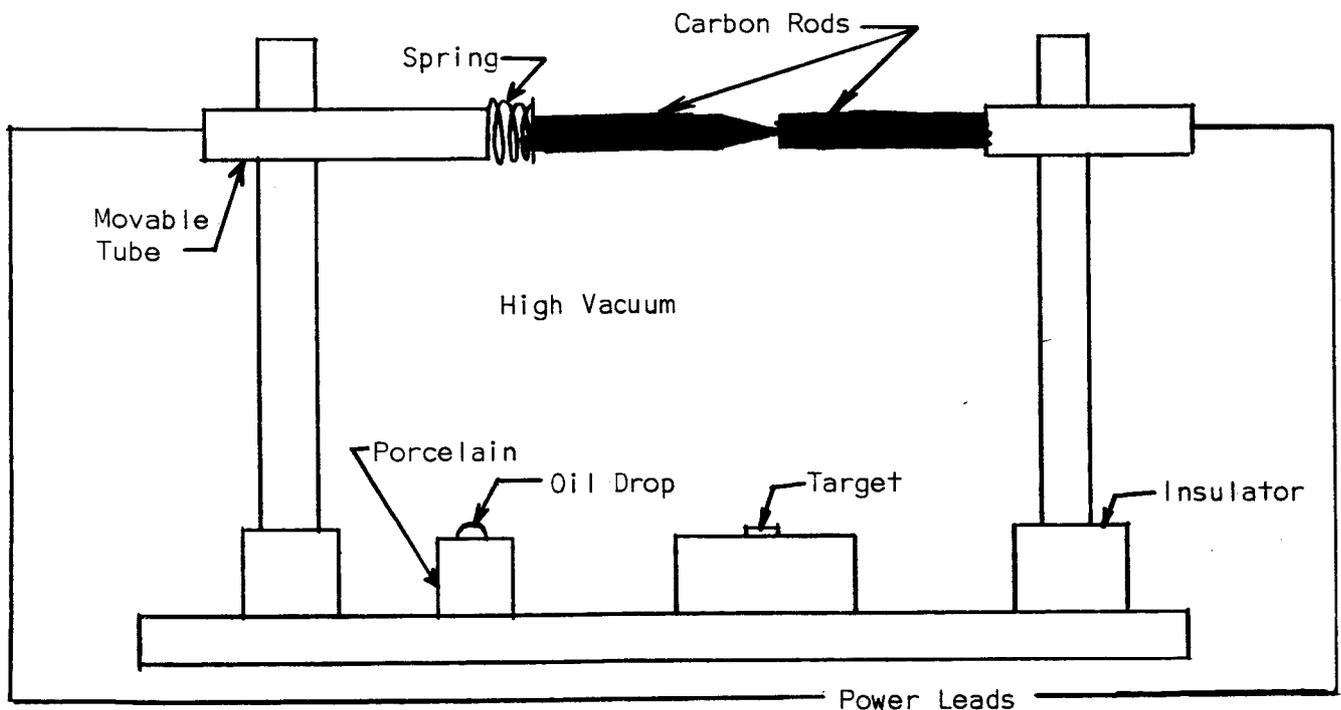


Figure 30  
Carbon Evaporation Unit

The unit used is the JEE-4B Vacuum Evaporator manufactured by Japan Electron Optics Laboratories Co., Ltd. The apparatus has the following features:

1. As the bell jar is furnished with two independent electrodes, one with a pair of filament holders and the other with a pair of filament holders and of carbon holders, two different kinds of metals can be evaporated successively, or carbon evaporation and metal evaporation can be performed at the same time or successively, by proper use of these holders.

2. The highly efficient oil diffusion pump and the rotary pump are used for keeping a vacuum less than  $10^{-5}$  mm Hg. Furthermore, pre-evacuation can be performed through the by-pass valve so that vacuum will be restored to the degree above immediately after exchange of a specimen.

The procedure is as follows: Two hard carbon rods are mounted in insulated holders. The blunt one is fixed and the pointed one is movable. A spring maintains a light pressure at the points. The target is placed about 10 cm from the source. A thickness indicator, consisting of a piece of white porcelain with a drop of vacuum oil on it is placed beside the target. The apparatus is covered with a bell jar and evacuated to between  $10^{-4}$  to  $10^{-5}$  torr. The carbon is now evaporated. The best thickness is believed to be obtained on this apparatus by passing 50 amps for a time of thirty seconds. The thickness is judged by the brown coloration on the indicator. This is visible on the area of the porcelain not covered with oil. When the brown color is just detectable compared to the white of the oil-covered area, the thickness of the film on the target is about  $50 \text{ \AA}$ . Other thicknesses will be experimented with to try to obtain the optimum. This would be the

thinnest carbon film which could be obtained and which would still hang together. Naturally the thicker the film, the poorer the resolution.

### Shadow Casting

Shadow casting is a technique which consists of depositing by vacuum evaporation a layer of electron-dense material onto the specimen at an angle.

(Figure 31)

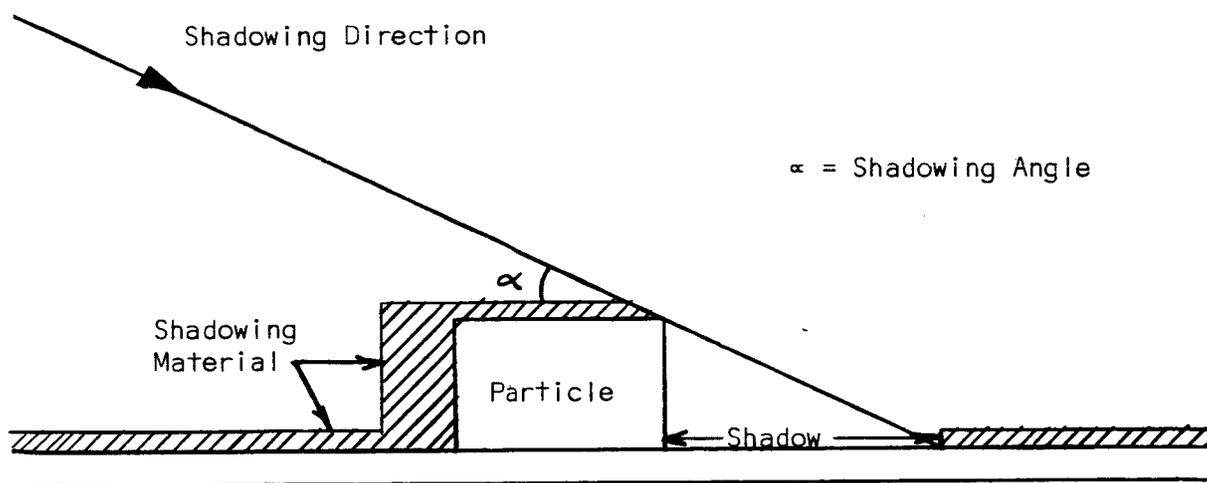


Figure 31  
Shadow Casting

The areas shielded from the impinging beam of atoms by surface irregularities are not coated. These areas are more electron-transparent than the coated areas and resemble shadows in appearance. When examined in the electron microscope, they will appear lighter than the surrounding areas. When a micrograph is taken of such a specimen the images of the shadows are reversed and appear dark, thus giving the impression of a surface illuminated

obliquely by white light. A print is then made which reverses this and gives the same picture observed in the microscope. The main purpose here is to provide added contrast to the final image produced by the electron microscope. Also by knowing the angle of deposition, it is possible to determine the height or size of surface projections or to determine their shape by the length of the shadows.

Up to this stage in the project platinum has been the shadowing material. This is accomplished very easily in the JEE system described previously. Shadowing has been done by evaporating a short piece (about 1 inch) of 8 mil platinum wire wrapped around a tungsten filament by heating to 36 amps for 30 seconds. This method works fine now, but later when very high magnification is required it is believed that the platinum will show a granular structure. This will be overcome by using platinum/carbon shadowing. This method should be capable of very high resolution and cause only very fine background structure. The platinum and carbon are evaporated simultaneously to form a deposit which has a high electron scattering power, but does not form discrete crystallites associated with evaporated films of heavy metals. The method depends on the use of rod type electrodes consisting of a mixture of platinum and carbon. However, it is believed that equally good results can be obtained by using carbon rods in a set-up like Figure 30 and wrapping platinum wire on one of the electrodes. The angle of incidence should be very small, about five degrees. This method will be put to use when a high degree of magnification is desired.

## V. Fundamentals of Photo Resist Materials

### A. Introduction

The silicon-silicon oxide interface properties in production devices are intimately related to photo resist portions of the manufacturing processes. Application of the fundamental information sought in this program will require understanding the influence of the photo resist processing. This section summarizes the results of an initial literature study of the chemistry and technology of photo resist materials.

### B. General Information

Two basic types of photo resist, a positive and a negative type, exist. The positive type produces a positive image of the original drawing upon development, where the negative type produces a negative image of the original drawing. Therefore, if a film coated substrate is to have its remaining film (after etching) identical to the inked lines of the drawing and if the foil is coated with a negative resist, then a negative of the inked drawing must be used to expose the photosensitive resist.

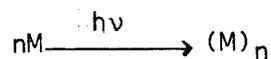
The most popular photosensitive resists in the electronics field are currently those products by the Eastman Kodak Co. (KMER, KTFR, KPR). Other resists (except AZ-1350, S-C grade from the Shipley Co.) are more commonly used or proposed to be used in various lithographic processes. Originally Kodak's photo resists were meant to revolutionize the printing industry; however, the electronics industry adopted the products. Now the three Kodak photo resist products, KPR, KMER, and KTFR (negative types), are the leading resists, respectively, in the copper foil coating applications, in the metal etch applications and the silicon shaping applications.

The positive photo resist available from the Shipley Co. may be satisfactory for metal etch operations (in particular semiconductor application) if it is desirable to eliminate the additional step of producing a negative of the original drawing. It would seem that further comments can be made only for specific cases since major semiconductor firms have chosen to remain with the negative photo resist processes. A major factor is the costly prospect of art work change-over.

The remaining photosensitive resist applications that appear in the literature seem to apply more to the printing trade. The basic spread of chemistry for these various photosensitive resists seems to be as follows: (1) the majority of these resists use some combination of polyvinyl alcohol (PVA), (2) Diazo Azide, Polyvinyl Cinnamate compounds, and phthalates of Hydroxy-containing polymers as the remaining basic building blocks.

Woodward, Chambers, and Cohen provide quite a useful insight((in their article, "Image-Forming Systems Based on Photopolymerization) into the chemistry and photographic characteristics of the photopolymerization. Some of the main points are as follows:

(1) Photopolymerization (first reference to it in 1945 by Gates) is outstanding as a result of its potential amplification, short access time, short chemical processing, stability, and versatility aspects. The spontaneous amplification of this widely studied system of photochemical image formation results in a high chemical quantum yield through a chain reaction in which a monomer, M, is converted to a polymer, (M)<sub>n</sub> in the following reaction:



where  $n$ , the kinetic chain length, roughly approximates the quantum yield.

(2) The speed of a photographic system is determined by its overall utilization of information bearing incident light to record and detect this recorded information. Very high speed systems might result if the quantum yields are on the order of  $10^6$  and  $10^9$  (i.e. a polymerization yield of  $10^6$  monomers per photon absorbed). As of 1963 the best that had been obtained was about  $10^3$ . However, chain lengths up to  $10^8$  are obtainable in addition polymerization. Addition polymerization does require special monomers, elimination of induction periods and rigorous control of inhibitors.

(3) Quite a bit of information is still lacking; such as the complete chemistry of photopolymerization (a) for excited states; (b) for mechanisms for energy transfer and radical formation, and (c) for the relationship of polymerization reactions to physical property changes.

(4) The primary chemical step is photoinitiation which occurs when an initiator component absorbs radiation at some wavelength between 2000 and 7000 Å to form an excited species capable of generating a polymerization-initiating free radical. This excited species (the same molecular structure, but now more energetic) as compared to the unexcited species is potentially capable of many reactions. The photon's energy causes a rearrangement of the electrons involved in the formation of chemical bonds and results in many paths by which the energy of such excited molecular states may be dissipated or utilized.

(5) The polymerization takes place when molecules at an excited state produce polymerization initiating radicals.

(6) An antihalation coating can decrease undesirable photopolymerization

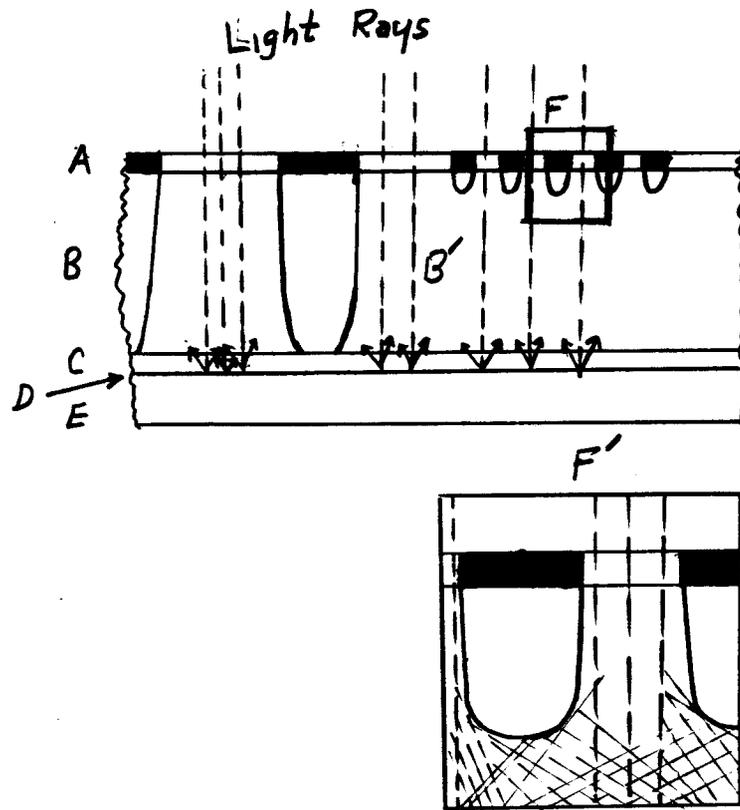


Figure 32  
Photopolymer Cross Section During Exposure

- (A) Film negative
- (B) Photopolymer layer, (B') Exposed (insoluble) polymer layer
- (C) Non-sensitive bonding layer
- (D) Anti-halation coating
- (E) Steel Support (substrate)
- (F & F') Section and cross section (underexposed)

since reflections otherwise blur the film negative image impressed upon the light-sensitive polymer. (See Fig. 32)

It is now possible to appreciate the general aspects of the presently existing photo resist materials.

### C. KMER and KTFR

#### 1. General Chemical Structure

These trademarked products of Eastman Kodak Company are more completely titled Kodak Metal Etch Resist and Kodak Thin Film Resist. Eastman Kodak Co. has not supplied any chemistry particulars other than application information, some of the latter dated as late as 1964, and as a result a considerable amount of literature search has been necessary in order to establish a reasonable basis for chemical understanding of KMER and KTFR. The details are now presented.

KTFR is a refinement of KMER. KMER has previously been "cleaned up" at least by a centrifuge process for high resolution applications. KMER may be thought to be very close to 3- and 4- ( $\alpha$ -cyanocinnamido) phthalates of hydroxyl-containing polymers. The following information came from personal communications, the Patent Gazette, and Chemical Abstracts.

The sections B and C, Fig. 33a and the sections B and C, Fig. 33b, can be identical. However, section A, Fig. 33, is polyvinyl cellulose. It is necessary to consider further the patent 2,861,057 information from the Gazette, the Chemical Abstracts, and personal communications. Since the patent information and other sources hint that section A is polyvinyl alcohol, KMER and KTFR may be supposed to have PVA as section A of the chemical structure (See Fig. 33). Since the patent 2,861,057 (filed December 17, 1956 and granted November 18, 1958) is the latest of the series

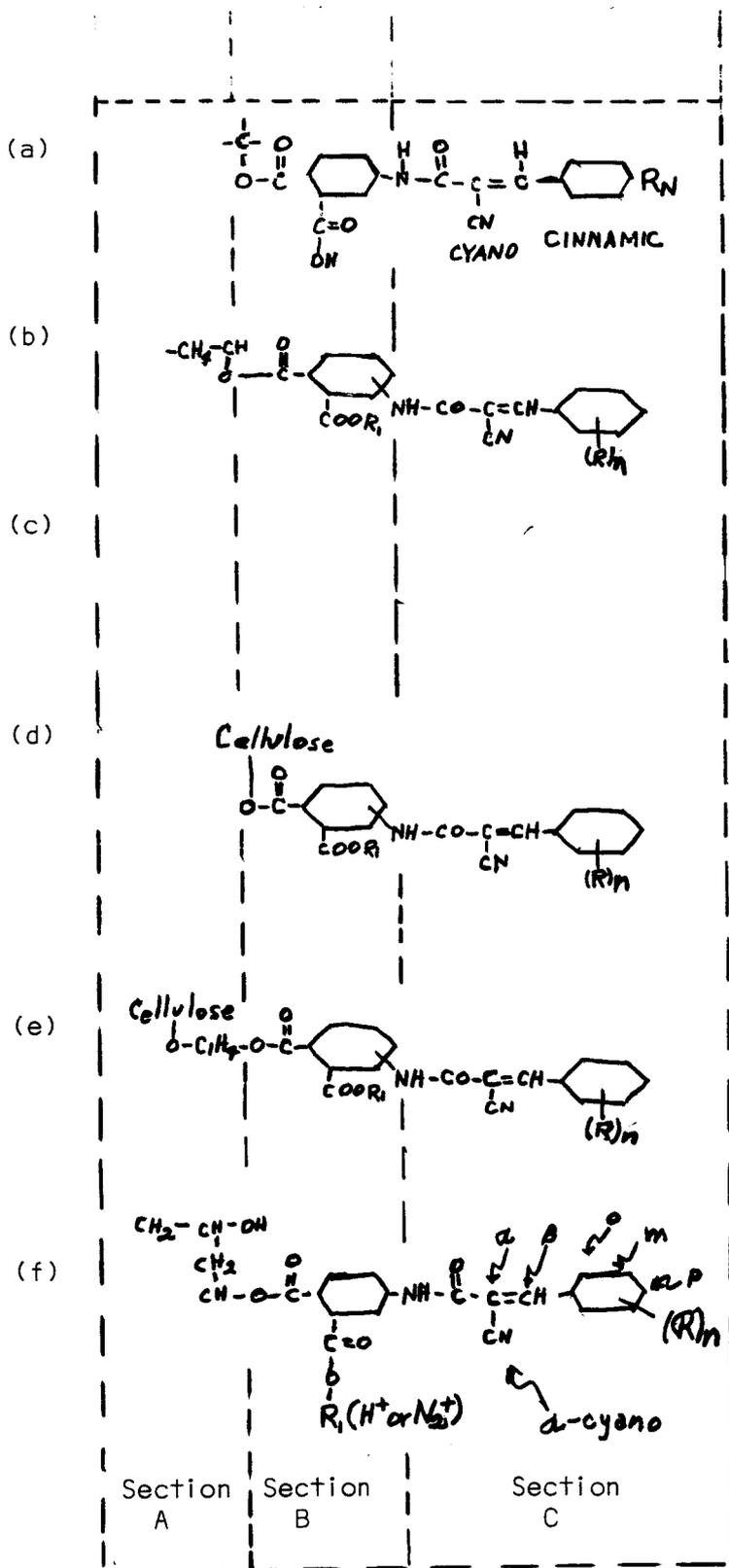


Figure 33  
Photosensitive Resist Chemical Chains

- (a) Partial KMER chain
- (b-e) Chains in Merrill Patent
- (f) Complete general KMER, KTRF chain

of light sensitive polymer patents:

2,835,656 filed November 25, 1953

2,848,328 and 2,852,379 filed May 4, 1955

2,861,058 filed July 29, 1955

2,865,750 filed March 18, 1955

(issued before the KMER trademark 752,504 and granted November, 1958, (trademark use date January 4, 1960, and filing date September 20, 1962), and it describes light sensitive polymers of 6,000 times the sensitivity of poly (vinyl cinnamate), this latter mentioned in patent application 2,865,750; then there exists a reasonable basis for belief that KMER and KTFR have the basic chain of Fig. 33 (originally described in patent 2,861,057).

The KMER, KTFR chemistry can be a variation of the basic patent since the take-off from the benzene rings are in a flexible configuration. This general configuration allows possible later refinements for more desirable characteristics.

## 2. Specific Chemical Structure

The general patent (14) comments about the structural chains are as follows:

"3- and 4- ( $\alpha$ -cyanocinnamido) Phthalates of Hydroxyl-containing polymers (can be) 1. A resinous polymer selected from the group consisting of (1) a vinyl polymer consisting of at least 30% by weight of the recurring general structural unit: (Fig. 33) wherein n represents an integer of from 1 to 2 and R represents a member selected from the group consisting of a hydrogen atom, a chlorine atom, a bromine atom, an alkyl group of 1 to 4 carbon atoms, an alkoxy group of 1 to 4 carbon atoms, an acetamido group, a

-COOR<sub>1</sub> group, an -SO<sub>3</sub>R<sub>1</sub> group and an (Fig. 33) group and wherein in each instance R<sub>1</sub> represents a member selected from the group consisting of a hydrogen atom and an alkali metal atom and R<sub>2</sub> represents an alkyl group of 1 to 4 carbon atoms, the remainder of the polymer molecule to make a total of 100% being composed of residual recurring structural units selected from the group consisting of a vinyl alcohol unit, a vinyl ester of a saturated monobasic fatty acid of 2 to 4 carbon atoms unit, and vinyl alcohol and said vinyl ester units in linear combination, (2) a cellulose ester consisting of at least 30% by weight of the recurring general structural unit: (Fig. 33) wherein n, R and R<sub>1</sub> are as above defined, the remainder of the cellulose ester molecule to make a total of 100% being cellulose acetate units, and (3) a cellulose ether consisting of at least 30% by weight of the recurring general structural unit: (Fig. 33) wherein n, R and R<sub>1</sub> are as above defined, the remainder of the cellulose ether molecule to make a total of 100% being hydroxyethylcellulose units."

The phthalate group (Fig. 33, section B) and α-cyanocinnamide group (Fig. 33, section C) may be attached either to Polyvinyl Cellulose or Polyvinyl Alcohol. The Chemical Abstractor for this patent outlines the procedure for the polyvinyl alcohol combination.

### 3. Properties

KMER and KTFR are essentially the same type of honey-colored, light sensitive resist. Both resists are low molecular weight compounds that use the same chemical processing techniques. KMER differs by the fact that it has an observable graininess in its KMER bottle (This graininess has caused its users to refer to it as "applesauce". The graininess supposedly

stems from a particular statistical frequency distribution of polymer length). The ash content of KTRF is kept under .0003% so that very high resolution is possible. KMER is recommended for thick coatings of resist since the exposure time for KTRF may be excessive (for a coating thickness of 0.0003 inch or greater). KMER single-coat thickness may range to 0.001 inch.

KMER was basically designed for aluminum, tool steel, stainless steel, titanium, and other structural metals; however, semiconductor use has increased its use to include germanium, molybdenum, tantalum, silicon, nickel, and (with additives) glass, and silicate surfaces. KMER can also be used for gold, magnesium, copper, nichrome, zinc, silver and plastics not attacked by aromatic hydrocarbons. KPR is preferred for copper.

KMER and KTRF will resist acids such as  $\text{HNO}_3$ ,  $\text{H}_2\text{SO}_4$ ,  $\text{H}_3\text{PO}_4$ , HF, some alkaline solutions such as NaOH, and water as a solvent. Hot chromate acid +  $\text{H}_2\text{SO}_4$  will remove the resists but other resist-strippers may be preferred or required such as Litho Aluminum Products Company's Resist-Strip J-100 or boiling  $\text{C}_2\text{HCl}_3$ . Chromate acid may adversely coat glass surfaces.

KTRF may adhere also to the same materials as KMER but the resolution requirement is the governing factor in the choice between KMER and KTRF resists. Heating of the resist should be limited to a temperature of 100°C. Adhesion will rise when heated until there is a color change from honey to milky-white. Past this point adhesion will drop to 10%. Excessive heating can cause a developed image to flow.

KMER, KTRF photo sensitivity extends into the visible light region so that an incandescent light source, even through certain glass,

will cause photopolymerization to take place.

Irradiation produces cross-linking so that the image remains upon development.

Note that for  $R_1$  (Fig. 33) being a  $N_a$  salt then the KMER, KTFR compound is basic and tends to be more water soluble. If  $R_1$  is a free acid ( $H^+$ ) then the KMER, KTFR compound tends to be more soluble in organic solvent.

#### 4. State-of-the-Art Techniques

Both resists must be applied to clean surfaces if undercutting due to poor adhesion is to be avoided. Sometimes if adhesion difficulty still prevails then either bake resist longer (at  $100^\circ C$ ) after development stage or (1) use Kodak's additive D (for glass and ceramic), (2) use a fluorocarbon non-ionic surface wetting agent (Minnesota Mining and Manufacturing Company's LI060) with the resist in silicon or silicon-oxide applications.

KTFR has a much higher resolution than KMER (KMER tends to fuss), the good resolution of KTFR permitting circuit elements of  $1/2$  micron separation. However, a limit of an element of 1 micron in width using KTFR seems to be the present case for one large semiconductor manufacturer. This limit of 1 micron seems to be a result of interference, bars and rings, that begin to appear; possibly as a result of the physical optics associated with the small apertures of the mask. Image reduction (via a lens system) during photo resist exposure is to be avoided where good overall replication is desired. Substrate indexing using a stereomicroscope (of at least 40X magnification) and a precise X-Y maneuverable chuck can give good replication as compared to the use of a fly's-eye lens system or a series of grid lines.

Humble Oil Company's Varsol works as a satisfactory substitute for Stoddard solvent developer. Use of alcohol instead of xylene prevents

accidental resist swelling. Resist coatings are best when a spin operation is employed to smooth the coating.

Addition of Stannous salts to photo polymerizable compositions decreases undercutting, and increases adhesion and image quality.

For good microcircuit replication every step must be completed under identical conditions. Masks may need to be from a drawing up to 500 times larger. Also, masks may need to be etched glass in order to maintain high dimensionable stability. Mask image must be adjacent to the substrate resist film and must be aligned and exposed by light rays having the same angle of incidence on the substrate plane ( $90^\circ$ ). Higher magnification with good definition can be obtained. A simple mask can also be made to produce several images.

The stripping of polymerized films (recommended by Keonjian) is by boiling in  $C_2HCl_3$  (trichloroethylene) then by "scrubbing" in acetone or heating in mineral acids such as  $HNO_3$  or  $H_2SO_4$ . Litho Products suggests their resist strip J-100.

#### D. Conclusion

The opportunities for extraneous materials to be included in a thin layer of oxide are numerous. Detailed analysis of surface chemical composition may be required to determine the influence of these materials on the properties of the  $Si/SiO_2$  interface.

## VI. Analytical Equipment

### A. Electron Mirror Microscope of Litton Industries

A visit to the Litton facility in Minneapolis revealed that the readily obtained maximum direct magnification with existing instruments is 3700X. For 0.37 mm as the minimum satisfactory spacing on the recording film, the minimum specimen resolution is  $10^{-7}$  m or 1000 Angstroms. Conversation with men working on the machine indicated that one to three years will be required for them to have studied the machine sufficiently thoroughly to have a definite indication of their capability on a project similar to the Si/SiO<sub>2</sub> interface study.

A few specimens should be submitted to them each year to determine if useful information can be obtained by the electron mirror approach.

### B. Fluorescence Analysis

Pursuit of analysis techniques for highly purified water may have led to an analysis technique with broad applications. The initial consideration was for a sensitive, convenient, and inexpensive indication of impurities in deionized water. The "ideal" approach being a device from which one can extract an electrical signal indicative of the impurity content. Such an electrical approach contrasts readily with the common techniques, usually requiring wet chemical analysis.

The electrical conductivity measurement on water gives the ionic conduction quite readily. Suspended inorganic solid materials and both dissolved and suspended organic materials do not contribute to ionic conduction. Other analytic techniques used include:

for Suspended Solids - residual mass after evaporation

- flow stoppage in a fine filter

for Organic Content - concentration, oxidation, and titration

- bacteria count

None of these are conveniently performed in anything less than a well equipped analytical laboratory. An electrical analytical technique offers much convenience.

Fluorescence in a material depends on the existence of certain relative energy levels for electrons around the atoms and appropriate electronic transition properties. Operationally, fluorescence is the excitation of material atoms at one wavelength of light with re-emission of radiation at another wavelength, usually a longer wavelength. Most organic materials fluoresce to some degree and a small fraction of inorganic compounds fluoresce. The best known examples of the latter are the natural minerals which fluoresce.

Research scientists in the life sciences have come to use the unique and powerful analytic capabilities of fluorescence phenomena. An extremely large number of organic molecules have unique fluorescence spectra just as some have unique infrared or ultraviolet absorption spectra. Because the kind of fluorescence spectrum of a material depends on the wavelength of the excitation radiation, fluorescence analysis has another dimension over the ordinary absorption spectrum techniques. Fluorescence can be an exceptionally powerful analytic phenomenon in many cases. For those materials which do not fluoresce, it obviously has no analytical value.

Deionized water systems can be installed and filtered to have

extremely low inorganic impurity content. The plumbing is organic, the deionizing resin is organic, and the most difficult of materials to remove from the raw water are the dissolved organics. Fractions of all these organics can be in the deionized water of a well-managed purification system. The concentrations are sufficiently low and the common analytical techniques so cumbersome that little is usually done to monitor this source of extraneous material in semiconductor device production.

A selected list of references has been examined to estimate how much information exists about the probable success of using fluorescence to monitor deionized water purity in conjunction with the usual ionic conductivity measurement:

Sidney Udenfriend, Fluorescence Assay in Biology and Medicine, Academic Press, 1962.

Gerald Oster and A. W. Pollister, eds., Physical Techniques in Biological Research, vol. III, Academic Press, 1956.

P. Pringsheim, Fluorescence and Phosphorescence, Interscience, 1949

C. A. Parker, "Raman Spectra in Spectroflurometry," The Analyst 84, 446 (1959).

Relatively few compounds have been thoroughly indexed for fluorescence analysis. Present research workers are contributing to a growing body of data.

A research investigation must be performed to isolate the primary organic constituents and the wavelength bands best for excitation and observation of the dominant impurities. Then a relatively simple filtered source exciter and filtered beam detector could give a quite meaningful relative indication of the quantity of organic material in the water and

whether the level is constant with time. Water has fluorescent bands of its own which prevent operation at those wavelengths, but this property offers an easy means of calibration for light source, optics, and electronics of the measurement system. A combination instrument for conductivity and fluorescence analysis can be compact and simple to use. The cost should be moderate. Such monitoring instruments can be put on each production line for indicating the quality of water used and the results of washing operations at critical steps in device manufacture. This appears to be a useful way of improving quality control on high reliability devices.

Instruments on the market can be readily adapted to such application.

Manufacturers include

American Instrument Company  
8030 Georgia Avenue  
Silver Spring, Maryland

G. K. Turner Associates  
Palo Alto, California

Farrand Optical Company  
Bronx Boulevard and East 28th Street  
New York 70, New York.

To obtain the information around which meaningful specifications for water could be written, a nine to twelve month program of water evaluation combined with device evaluation should be executed. A double monochromator instrument (with recorder) for precise analysis costs about \$7,000. An instrument using interference filters as the monochromators can be purchased for less than \$2,000. Use of these instruments in conjunction with other device research should produce very useful quality control specifications.

APPENDIX

Table 1

100 cps	<u>Volts</u>	<u>Capacitance</u>	<u>Conductance</u>
	0.10	116.10	0.0154
	0.30	93.18	0.0069
	0.50	79.48	0.0055
	1.00	61.37	0.0035
	3.00	40.43	0.0012
	5.00	29.30	0.0011
	10.0	22.20	0.0007
	30.0		
	50.0		
		unable to get indication	

Table 2

1000 cps	<u>Volts</u>	<u>Capacitance</u>	<u>Conductance</u>
	0.10	131.46	0.0960
	0.30	87.26	0.0366
	0.50	75.20	0.0267
	1.00	59.05	0.0156
	3.00	37.30	0.0057
	5.00	29.81	0.0037
	10.0	21.91	0.0027
	30.0	13.40	0.0022
	50.0	10.20	0.0019

Table 3

10,000 cps	<u>Volts</u>	<u>Capacitance</u>	<u>Conductance</u>
	0.20	88.38	0.0110
	0.30	81.30	0.0100
	0.50	70.02	0.0087
	1.00	56.21	0.0073
	3.00	36.30	0.0063
	30.0	11.14	0.005
	50.0	10.10	0.005
	100.0	10.09	0.005

## BIBLIOGRAPHY

1. B. E. Deal, A. S. Grove, E. H. Snow, C. T. Sah,  
"Recent Advances in the Understanding of the Metal-Oxide-Silicon System,"  
Presented at the AIME Conference on Electronic Materials, Boston, September 1, 1964.
2. J. E. Thomas, Jr. and D. R. Young,  
"Space-Charge Model for Surface Potential Shifts in Silicon Passivated with Thin Insulating Layers,"  
IBM Journal of Research and Development, Vol: 8, No. 4, pp. 368-375, September, 1964.
3. H. H. Skilling,  
Electrical Engineering Circuits, New York, John Wiley & Sons, pp. 18-20, 1959.
4. L. P. Huselman,  
Circuits, Matrices, and Linear Vector Spaces, New York, McGraw-Hill Book Co., Inc., pp. 3-5, 112-113, 1963.
5. D. N. Chorofas,  
Statistical Processes and Reliability Engineering, New York, D. Van Nostrand Co., Inc., pp. 11-14, 1960.
6. "Operating Instructions Type 1615A Capacitance Bridge,"  
General Radio Company, West Concord, Massachusetts, October, 1963.
7. Personal Communication,  
Stan Gessel, Product Specialist, Electro-scientific Industries to Dr. H. Lyndon Taylor.
8. "Instruction Manual Electro-scientific Industries Model 277 Capacitance Bridge,"  
Electro-scientific Industries, Portland, Oregon, April, 1963.
9. P. J. Holmes,  
The Electrochemistry of Semiconductors, New York, Academic Press, 1962.
10. Lee Barnes,  
Private communication.
11. Karl H. Kretzer,  
"Positive Working Photoresist Layer,"  
Chemical Abstracts, Vol. 60, No. 2, p. 1265, January, 1964.

12. Andre K. Schnerin and H. D. Evans,  
"Photographic Polymeric Resist Images,"  
Chemical Abstracts, Vol. 60, No. 2, p. 1266g, January, 1964.
13. P. W. Woodward, V. C. Chambers, A. B. Cohen,  
"Image-Forming Systems Based on Photopolymerization,"  
Photo. Science Engineering, Vol. 7, No. 6, pp. 360-368, 1963.
14. Stewart H. Merrill and Donald A. Smith,  
"Light-Sensitive 3- and 4- ( $\alpha$ -Cyanocinnamoylamino) Phthalates of  
Hydroxy-Containing Polymers,"  
Chemical Abstracts, Vol. 53, p. 4989c, March, 1959.
15. W. E. Rowe,  
"Photographic Fabrication of Semiconductor Devices,"  
Chemical Abstracts, Vol. 58, p. 12069a, June, 1963.
16. Cornelius C. Unruh and D. A. Smith,  
"Light-Sensitive Polymers for Photography,"  
Chemical Abstracts, Vol. 53, 1959.
17. Cornelius C. Unruh, G. W. Leubner, A. C. Smith, Jr.,  
"Light-Sensitive Polymer for Printing Plates,"  
Chemical Abstracts, Vol. 53, p. 4991e, March, 1959.
18. E. Keonjian (editor),  
"Microelectronics, Theory Design, Fabrication,"  
McGraw-Hill, New York, New York, pp. 184, 177-178, 290-296, 1963.
19. Eastman Kodak Co., Rochester, New York,  
Industrial Data Handbook, p. 7, 1962, 1964.

## VII. General Bibliography Section

This bibliography was assembled from the NASA Scientific and Technical Aerospace Reports (STAR) Abstract Journal. This section is an addition to those in the previous report.

Bibliography  
on the  
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Huntsville, Alabama

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N63-10183-01-15, AMRL-TDR-62-88,

M. J. Allen,

"Calibration of the Infrared Optometer. Technical Documentary Report (May 1, 1961 - March 13, 1962),"

August, 1962, Contract AF33(616)-6146

N63-10282-01-18, DMIC-MEMO-160,

C. H. Lund, H. J. Wagner,

"Identification of Microconstituents in Superalloys,"

November 15, 1962, Contract AF33(616)-7747

N63-10367-01-25, AROD-2838-1,

A. M. Goodman, W. W. Mehl,

"Research on Metallic Contacts to Semiconductors. Final Report for the Period June 23, 1960 to June 22, 1962,"

July 26, 1962, Contract GA-034-ORD-3282RD

N63-10410-01-09, NBS-TN-201,

R. N. Jones,

"A Technique for Extrapolating the I<sub>KC</sub> Values of Secondary Capacitance Standards to Higher Frequencies,"

November, 1963

N63-10498-01-23,

A. M. Russel,

"Tables of Non-Relativistic Electron Trajectories for Field Emission Cathodes,"

October, 1962, Contract Nonr-1842(02)

- N63-10507-01-25, AFCRL-62-772,  
M. M. Atalla, R. W. Soshea, R. C. Lucas, C. H. Fox, V. M. Dowles,  
"Investigation of Hot Electrons Emitter. Science Report No. 1,  
June 1 - August 31, 1962,"  
1962, Contract AF19(628)-1637
- N63-10562-01-15, AFCRL-62-1070,  
E. N. Lassetre, F. M. Glaser, V. D. Meyer, A. Skerbele,  
"Recent Advances in the Use of the Electron Spectrometer,"  
September, 1962, Contract AF19(604)-4541
- N63-10674-02-15, AEC-TR-5080,  
Heinz Ewald and Heimich Hintenberger,  
"Methods and Uses of Mass Spectroscopy,"  
July, 1962
- N63-10759-02-25, AFCRL-62-585,  
M. K. Wilson,  
"Spectroscopic Investigations of Solid State Surfaces Containing  
Absorbed Molecules. Final Report,"  
May, 1962, Contract AF19(604)-4063
- N63-10876-02-16, ARL/MET-15,  
T. Mills,  
"A Vacuum Microbalance for High Temperature Oxidation Studies.  
Metallurgy Note 15,"  
April, 1962
- N63-11173-03-23, AFCRL-62-904,  
O. Heil and S. Vogel,  
"Surface Explosions on Solids by High-Density Electron Bombardment,"  
October, 1962, Contract AF19(628)-1606
- N63-11239-03-01, WACC-TK-53-373, Supplement 9,  
D. J. Tate,  
"A Review of the Air Force Materials Research and Development  
Program. Final Report, July 1, 1961 - June 30, 1962,"  
November, 1962
- N63-11691-04-15, BM-RI-6111,  
J. D. Brown, et. al.,  
"Design, Construction, and Evaluation of an Electron Probe X-Ray  
Spectrograph,"  
1962
- N63-11803-04-15, AFCRL-62-933,  
T. O. Passell, J. B. Swediunst, et. al.,  
"Development of X-Ray Fluorescence Analysis Techniques for Small  
Samples: A Prototype Atmospheric Electron Probe - Final Report,"  
October 31, 1962, Contract AF19(604)-8827

- N63-11843-04-07, JPRS-17176,  
L. A. Gribov,  
"Methods of Calculation in Molecular Spectroscopy,"  
January 17, 1963
- N63-11952-05-09,  
D. R. Chambers,  
"Integration of Energy Storage and Discharge Mechanisms in Bulk Silicon and Germanium for Application in Neuristor Lines. Interim Report 6,"  
October, 1962, Contract Nonr-3112(00) and AF33(657)-7801
- N63-12094-05-23, MRL-126,  
A. N. Goland,  
"The Role of Electron Microscopy in Observations of Radiation Damage,"  
November, 1962
- N63-12184-05-23, DMIC-MEMO-161,  
W. R. Warke and A. R. Elsea,  
"Electron Microscopic Fractography,"  
December, 1962, Contract AF33(616)-7747
- N63-12579-06-23,  
R. W. G. Wyckoff, F. D. Davidson,  
"Windowless Tubes for X-Ray Spectroscopy,"  
1962, NASA Grant NSG-120-61
- N63-12643-06-25, AFCRL-62-949,  
H. E. Farnsworth, J. A. Dillon, Jr., et. al.,  
"Investigations of Surface Properties of Silicon and Other Semiconductors. Final Report, October, 1959 - September 30, 1962,"  
November, 1962, Contract AF19(604)-5986
- N63-12945-07-09,  
P. K. Weimer, H. Borkan, et. al.,  
"Evaporated Thin Film Devices. Scientific Report No. 1, Period June 1, 1962 to November 30, 1962,"  
November, 1962
- N63-12988-07-23,  
F. D. Davidson, R. W. G. Wyckoff,  
"X-Ray Fluorescent Yields from Several Light Elements,"  
December, 1962, NASA Grant NSG-120-61
- N63-13053-07-23, AROD-2588-3,  
M. P. Givens,  
"A Study of the Optical Properties of Solids in the Extreme Vacuum Ultraviolet and/or the Soft X-Ray Regions. Final Report, June 1, 1960 - October 31, 1962,"  
January 8, 1963, Grant DA-ORD-17

- N63-13191-07-01,  
George C. White, Jr., et. al.,  
"Basic Research in Physical Sciences. Second Quarter Progress Report  
for PY 63,"  
December, 1962
- N63-13213-07-01,  
C. D. Thibault, ed.,  
"Materials Research Abstracts, A Review of the Air Force Materials  
Research and Development,"  
1962
- N63-13593-08-15,  
A. M. Russell,  
"Field Emission Spectrometer, Technical Report No. 5,"  
December, 1962, Contract Nonr-1842(02)
- N63-13796-09-23, AEDC-TDR-63-49,  
R. L. Heimburg, et. al.,  
"Measurement of Pressure in an Ultra-High Vacuum,"  
March, 1963, Contract AF40(600)-909
- N63-13599-08-23,  
"Third International Symposium on X-Ray Optics and X-Ray Microanalysis,  
Stanford University, August 22-24, 1962,"  
Grant NSG-296-62
- N63-14255-10-15, BM-R1-6212,  
F. D. Stevenson, C. E. Wicks,  
"A Metal Diaphragm Apparatus for Measuring Vapor Pressure. Vapor  
Pressure of Arsenic (III) Oxide,"  
1963
- N63-14276-10-15, NASA TND-1695,  
M. M. Sokoloski, F. E. Geiger,  
"Apparatus for the Measurement of Transport Properties of Polar  
Semiconductors,"  
1963
- N63-14294-10-18, USBM-U-1001,  
A. H. Roberson, et. al.,  
"Quarterly Metallurgical Progress Report No. 17 for the Period of  
October 1 through December 31, 1962,"  
Contract AT(11-1)599
- N63-14412-10-25, AEC-TR-4475,  
S. I. Pekar,  
"Research in Electron Theory of Crystals,"  
February, 1963

- N63-14533-10-09, AFCRL-63-21,  
H. W. Cooper, et. al.,  
"Research and Development for Surface Protection for Silicon Devices,  
Scientific Report No. 3, June 15 - September 14, 1962,"  
October, 1962, Contract AF19(604)-8358
- N63-15161-11-25,  
W. J. Fredericks, F. J. Keneshea, F. Craig,  
"A Bibliography of Group I-VII Elements and Their Compounds, Volume I  
Scientific Report No. 2,"  
1946-1961, Contract AF19(604)-7231
- N63-15162-11-25,  
W. J. Fredericks, F. J. Keneshea, F. Craig,  
"A Bibliography of Group II-VI Elements and Their Compounds, Volume II  
Scientific Report No. 2,"  
1946-1961, Contract AF19(604)-7231
- N63-15163-11-25,  
W. J. Fredericks, F. J. Keneshea, F. Craig,  
"A Bibliography of Group III-V Elements and Their Compounds, Volume III  
Scientific Report No. 2,"  
1946-1961, Contract AF19(604)-7231
- N63-15164-11-25, AFCRL-62-917(IV),  
W. J. Fredericks, F. J. Keneshea, F. Craig,  
"A Bibliography of Group IV Elements and Their Compounds, Volume IV  
Scientific Report No. 2,"
- N63-15802-13-25, NOLTR-62-125,  
D. F. Bleil, et. al.,  
"Solid State Research of the Applied Physics Department for the Year  
1961,"  
July 16, 1962
- N63-15960-13-20, LMSC-6-90-63-23,  
H. P. Greinel,  
"Diffraction Pattern Analysis of a Rectangular Aperture in the  
Presence of Aberrations Part 2: The Diffraction Pattern in the  
Presence of Single Third-Order Aberrations: 1. The Effect of  
Primary Astigmatism,"  
May, 1963
- N63-16106-13-25, RADC-TDR-63-30,  
S. M. Skinner, J. W. Dzmianski,  
"Study of Failure Mechanisms. Final Report, July 1, 1961 to  
September 30, 1962,"  
December, 1962, Contract AF30(602)2558

- N63-16144-13-07, UCRL-7238,  
J. E. Senkin and M. I. Milliken,  
"Absorption Spectra of Various Metallic Halides and Oxides,  
Elemental Carbon and the Inert Gases: A Bibliography,"  
February, 1963, Contract W-7405-eng-48
- N63-16978-14-23, BM-1C-8166,  
Howard M. Stainer,  
"X-Ray Mass Absorption Coefficients, A Literature Survey,"  
1963
- N63-17083-15-25, AFCRL-62-949, Supplement,  
J. A. Dillon, Jr. and R. M. Oman,  
"Investigations of Surface Properties of Silicon and Other Semi-  
conductors. Final Report, 30 September, 1962 - 31 January, 1963,"  
February, 1963, Contract AF19(604)-5986
- N63-17351-15-09, ONR-14,  
"Proceedings of Second Conference on the Navy Microelectronics Program,  
Washington D.C., September 24 - 26, 1962,"  
1963
- N63-17388-15-25, AFCRL-63-113,  
"Investigations of Hot Electron Emitter - Scientific Report No. 3,  
December 1, 1962 - February 28, 1963,"  
Contract AF19(628)-1637
- N63-17608-16-25, AFCRL-63-65,  
F. Euler and G. H. Schwuttke,  
"Dislocation Studies in Diamond By X-Ray Diffraction Microscopy,"  
March, 1963
- N63-17902-16-15, AID-T-63-86,  
A. I. Akishin, V. S. Zozulin,  
"Film Thickness Control by Means of a Crystal Resonator,"  
1963
- N63-18400-17-25, NASA CR-50456,  
F. R. Brotzen,  
"Physics of Solid Materials. Semiannual Status Report, November 1,  
1962 - April 30, 1963,"  
NASA Grant NSG-6-59
- N63-18640-17-09, RADC-TDR-63-152,  
K. Greenough, F. Reed, and N. Johnson,  
"Study of Failure Mechanisms at Surfaces and Interfaces,"  
March, 1963, Contract AF30(602)-2593

- N63-18853-18-25, ORNL-3364,  
"Solid State Division Annual Progress Report for Period Ending August  
31, 1962,"  
November 15, 1962, Contract W-7406-eng-26
- N63-19432-19-09,  
C. E. Horton, J. W. Hall, II,  
"Research, Development, and Fabrication of Tunnel Emission Cathodes.  
Second Quarterly Interim Technical Report, October - December, 1962,"  
1962, Contract DA-49-186-OKD-105
- N63-19713-19-18,  
K. Lucke,  
"Recrystallization of Single Crystals. Third Quarterly Technical  
Status Report, December 1, 1962 - February 28, 1963,"  
Contract DA-91-591-EUC-2438
- N63-19717-19-18,  
P. P. Dennis and J. S. Brammar,  
"The Preparation of Thin Metal Foils by Microtomy: An Assessment of  
the Technique and its Application to Metallurgical Investigation  
with the Electron Microscope. Final Technical Report, December 31,  
1961 - March 30, 1962,"  
April 22, 1963, Contract DA-91-591-EUC-1695
- N63-19805-19-25, AROD-1176-20,  
G. A. Jeffrey,  
"Research and Development of Electron Density Distribution in Semi-  
conductors. Final Technical Report, January 1, 1961 - December 31,  
1962,"  
Grant DA-ORD-31-121-61-G63
- N63-20122-20-09, AERE-M-903,  
A. R. Owens,  
"The Behavior of Semiconductor Diodes in Pulse Applications,"  
January, 1962
- N63-20-145-20-15, AROD-3975-1,  
Grahm Gurr,  
"A Normal Beam Automatic X-Ray Diffractometer,"  
June 20, 1963, Contract DA-ARO(D)-31-124-G412
- N63-20230-20-25, ORNL-3480,  
"Solid State Division Annual Progress Report for the Period Ending  
May 31, 1963,"  
August 23, 1963, Contract W-7405-eng-26
- N63-20267-20-18, DMIC-MEMO-161,  
W. R. Warke, A. R. Elsea,  
"Electron Microscopic Fractography,"  
December 21, 1962, Contract AF33(616)-7747

N63-20283-20-25, DS-117,  
M. Neuberger,  
"Silicon: Absorption Data Sheets,"  
December, 1962, Contract AF33(616)-8438

N63-20287-20-25, DS-120,  
M. Neuberger,  
"Silicon: Mean Free Path Data Sheets,"  
January, 1963, Contract AF33(616)-8438

N63-20289-20-25, DS-126,  
M. Neuberger,  
"Silicon: Electrical Conductivity Data Sheets,"  
June, 1963, Contract AF33(616)-8438

N63-20300-20-25, DS-119,  
M. Neuberger,  
"Silicon: Dielectric Constant Data Sheets,"  
January, 1963, Contract AF33(616)-8438

N63-20375-20-25, DS-118,  
M. Neuberger,  
"Silicon: Debye Temperature Data Sheets,"  
January, 1963, Contract AF33(616)-8438

N63-20717-21-09, NYU-TR400-81, AFCRL-63-317,  
G. J. Herskowitz,  
"Some Network Properties of the Large Area PN Junction, Fourth  
Scientific Report,"  
July, 1963, Contract AF19(628)-379

N63-20719-21-25, AFCRL-63-815,  
George Rupprecht,  
"The Surface Barrier Effect in Photovoltaic Conversion. Final  
Report,"  
May, 1963, Contract AF19(604)-8506

N63-21789-22-25, AMRA-TR63-03, MRL-130,  
D. R. Chipman, B. W. Buttermann,  
"Contribution of Thermal Diffuse Scattering to Integrated Bragg  
Reflections from Perfect Crystals,"  
January, 1963

N63-21806-22-23, LMSC-6-90-62-35, SRB-62-41,  
J. B. Goldman,  
"Small Angle X-Ray Scattering: An Annotated Bibliography,"  
June, 1962

- N63-22081-22-09, AFCRL-63-336,  
M. M. Atalla, et. al.,  
"Investigation of Hot Electron Emitter, Scientific Report No. 4,  
March 1, 1963 - May 31, 1963,"  
1963
- N63-20205-20-25, AID-B-63-101,  
"Lattice Dynamics. Annotated Bibliography of Soviet Literature, 1962,"  
August 2, 1963
- N63-22276-23-23, RFP-326,  
G. J. Werkema,  
"DRJXR.1: A Fortran Computer Program to Calculate Crystallographic  
D-Spacings and Bragg Angles,"
- N63-22547-23-09, RADC-TDR-63-326,  
Hans J. Queisser,  
"Failure Mechanisms in Silicon Semiconductors,"  
August, 1963, Contract AF30(602)-3016
- N63-22559-23-23, RSIC-65,  
J. G. Helmaks,  
"Theory and Practice of Electron Microscopic Stereophotography,"  
September 13, 1963
- N63-22576-23-07, LMSC-SB-63-32,  
H. M. Abbatt, J. V. Kerrigan,  
"Epitaxial Growth: An Annotated Bibliography,"  
April, 1963
- N63-22632-23-09, Q445-3,  
Egons Rasmanis, James Cline,  
"Active Thin-Film Techniques Micromin Program. Third Interim  
Development Report, December 23, 1962 - March 22, 1963,"  
April 14, 1963, Contract NObsr 87633
- N63-22633-23-09, Q445-4,  
Egons Rasmanis, James Cline,  
"Active Thin-Film Techniques Micromin Program. Fourth Interim  
Development Report, March 23 - July 22, 1963,"  
August 20, 1963, Contract NObsr 87633
- N63-22789-23-09,  
"Semiconductor Thin Films. Quarterly Report No. 6, April 1 - June 30,  
1963,"  
Contract AF33(657)-7623
- N63-23084-24-25,  
"Research on Semiconductor Transport. Annual Summary Report, October 1,  
1962 - September 30, 1963,"  
Contract Nonr 2934(00)

N63-23353-24-01, SID-63-167,  
"Ultrasonic Cleaning; A Bibliography,"  
March 1, 1963

N63-23061-24-25,  
G. H. Wannier,  
"Dynamics of Block Electrons in Solids,"  
October 1, 1962, Contract Nonr 2771(05)

- N64-10096-01-09, NASA CR-52190,  
S. A. Ring, J. B. Story,  
"Electrolytic Preparation of High Dielectric Thin Films. Quarterly  
Report, June 27, 1963 - September 27, 1963,"  
1963, Contract NASw-765
- N64-10257-01-01, IS-700,  
"Annual Summary Research Report of Chemical Engineering, Metallurgy,  
Physics and Reactor Divisions. Annual Summary Report, July 1, 1962 -  
June 30, 1963,"  
September, 1963, Contract W-7405-eng-82
- N64-10307-01-23,  
A. E. Spakowski,  
"Use of Polaroid Film in X-Ray Diffractions,"  
August, 1963
- N64-10622-01-09, LS-BIB-63-3,  
C. L. M. Blocker, M. Bloomfield, T. J. Taylor,  
"Epitaxial Deposition of Semiconductor Materials: An Annotated  
Bibliography,"  
October, 1963
- N64-10705-02-18, TRRS-17638,  
"Soviet Research in Rare Metal Industry,"  
February 14, 1963
- N64-10708-02-25, JPRS-17689,  
"Translations from Fizika Tverdogo Tela (Solid State Physics)(Leningrad,  
V. 4) No. 7, 1962,"  
February 18, 1963
- N64-10732-02-25, JPRS-18008,  
"Translations from Fizika Tverdogo Tela (Solid State Physics) No. 9,  
1962,"  
March 7, 1963
- N64-10737-02-23, JPRS-18012,  
L. Belgagio,  
"Crystallography and Technology,"  
March, 1963
- N64-10738-02-25, JPRS-18062,  
"Translations from Fizika Tverdogo Tela (Solid State Physics) No. 10,  
1962,"  
March 11, 1963
- N64-11028-02-25, AD-423328,  
"Semiconductor Thin Films. Quarterly Report No. 7, July 1 - September 30,  
1963,"  
1963, Contract AF33(657)-7623

- N64-11466-03-18,  
G. V. Samsonov, V. M. Sleptsov,  
"Preliminary Variant of the Phase Diagram of the Boron-Silicon  
System,"  
November 13, 1963
- N64-11561-03-23, NASA-TTF-73,  
L. N. Dobretsov,  
"Electron and Ion Emission,"
- N64-11923-03-15, JPRS-22443,  
Ye Ye Levin, V. M. Lukyanovich,  
"All-Union Conferences on Electron Microscopy and Nondestructive  
Methods of Testing,"  
December 24, 1963
- N64-11933-03-25, AFCRL63-333, AD422580,  
S. S. Flaschen, H. W. Cooper, et. al.,  
"Research and Development for Surface Protection for Silicon Devices.  
Scientific Report No. 4, 15 December 1962 - 14 March 1963,"  
Contract AF19(604)-8358
- N64-12095-03-07, EDR-3687,  
Charles Jirrell,  
"Survey of Applicable Electrochemical Technology, Operational and  
Fabrication Techniques and Materials,"  
December, 1963, Contract AT(30-1)-3101
- N64-12237-03-15, AFCRL63-350,  
D. M. Jackson, Jr.,  
"Epitaxial Control System,"  
January 14, 1963, Contract AF19(604)-8351
- N64-12590-04-25, RADC-TDR-63-443, AD-425472,  
H. Queisser, W. Shroen, et. al.,  
"Failure Mechanisms in Silicon Semiconductors,"  
November, 1963, Contract AF30(602)-3016
- N64-12826-04-09, AD-422096,  
A. R. Von Hippel,  
"Progress Report No. XXXIII, Laboratory for Insulation Research,"  
July, 1963, Contracts Nonr-184(10), Nonr-184(88), AF19(628)-395,  
AF33(616)-8353, and AF19(604)-8483
- N64-12998-04-09, AD-422536,  
W. Corrigan, et. al.,  
"Production Development of a Silicon Planar Epitaxial Transistor with  
a Maximum Operating Failure Rate of 0.001% per 1000 Hours at a Confi-  
dence Level of 90% at 25°C. Fourth Quarterly Report, February 1 -  
April 30, 1963,"  
Contract DA-36-039-SC86728

- N64-13031-04-09, AD-424628,  
J. M. Blank, A. E. Cahill, et. al.,  
"Research on Thin Film Insulators. Second Interim Engineering  
Report,"  
October 25, 1963, Contract AF33(657)-11079
- N64-13036-04-18, AD-423329,  
L. C. Kravitz,  
"ESR and Optical Experiments on Deep-Lying Impurities in Germanium  
and Silicon,"  
June 14, 1963, Contract Nonr-1866(10), Nonr-1866(16)
- N64-13038-04-09,  
F. Tumbelty,  
"Production Engineering Measure of 2N 1708 Silicon Planar Epitaxial  
Transistor. Quarterly Report No. 5, May 1 - July 30, 1963,"  
Contract DA-36-039-SC-86729
- N64-13075-04-09, AD-423673,  
S. J. Buginas,  
"Vacuum Disposition of Thin Film Electronic Components: An  
Annotated Bibliography,"  
February, 1963, Contract N0w-63-0050
- N64-13636-05-25, ARL63-197, AD426752,  
E. L. Kern, D. E. Floyd, P. Bunce,  
"Low Energy Electron Diffraction from Single Crystal Surfaces I.  
Apparatus Design,"  
November, 1963, Contract AF33(657)-10798
- N64-13784-05-25, RADC-TDR-63-464, AD-427117,  
G. Rapprecht,  
"Study of Surface States in Semiconductors,"  
December, 1963, Contract AF30(602)-3157
- N64-13785-05-25, RADC-TDR-63-474, AD-427092,  
K. K. Beinhart, V. A. Russell, et. al.,  
"Failure Mechanisms at Surfaces and Interfaces. First Interim  
Technical Documentary Report, July 1 - October 1, 1963,"  
December, 1963, Contract AF30(602)-3085
- N64-13961-05-25, AD-427339,  
M. Lavik,  
"The Structure and Strength of Epitaxial Films. Bi-monthly Progress  
Report No. 1, October 17 - December 17, 1963,"  
January, 1964, Contract N0w-64-0061-d

- N64-14019-05-07, AD-426742,  
Masey Bateman, et. al.,  
"Bibliography on Analytical Chemistry Relating to Borides, Carbides,  
and Nitrides of Group IV and V Elements,"  
1963, Contract DA-30-069-ORD-2787
- N64-14022-05-25, AD-401398,  
H. C. Gatos, M. C. Lavine,  
"Chemical Behavior of Semiconductors: Etching Characteristics,"  
January 2, 1963, Contract AF19(628)-500-APRA-SD90
- N64-14182-06-01, NAS-NRC-1139, AROD-3935.3, AD-424943,  
Daniel Berg, J. C. Devins,  
"Dielectrics: Digest of Literature, Vol. 26,"  
1962
- N64-14295-06-25, RTD-TDR-63-4125, AD-427178,  
W. J. King, et. al.,  
"P-N Junction Formation Techniques, Third Quarterly Report, May 20 -  
September 1, 1963,"  
November, 1963, Contract AF33(657)-10505
- N64-14504-06-09, AD-420958,  
R. M. Lichtenstein,  
"Investigation of Transient Radiation Damage in Semiconductor  
Materials,"  
1963, Contract DA-36-039-AMC-02165(E)
- N64-14685-06-23, AFCRL-63-388, AD-428785,  
John O'Connor,  
"Analysis of Impurities in Solid State Electronic Materials, Final  
Report,"  
October 1, 1963, Contract AF19(604)-7463
- N64-15339-07-07, ORNL-3537,  
M. T. Kelly, C. D. Susano,  
"Analytical Chemistry Division Annual Progress Report for Period  
Ending November 15, 1963,"  
February 18, 1964, Contract W-7405-eng-26
- N64-15563-07-01,  
W. W. Blender,  
"Research Institute For Advanced Studies, 1962, Summary of Year's  
Activities,"  
1962
- N64-15687-07-07, SSD-TDR-63-349, TDR-269(4210-10)3, AD-429199,  
P. Breisache,  
"Mass Spectrometry and the Properties of Condensed Systems at High  
Temperatures,"  
December 27, 1963, Contract AF04-(695)-269

N64-16033-07-07, NASA CR-55774,  
 "Solids Mass Spectrometer - First Quarterly Technical Progress Report,  
 November 4, 1963 - February 3, 1964,"  
 1964, NASA Contract NASn-839

N64-16216-08-23, JPRS-23510, OTS-64-21710,  
 "X-Ray Diffraction Research,"  
 March 4, 1964

N64-16434-08-23, ARL-63-246, AD-429228,  
 Per-olov Lowdin,  
 "Quantum Theory of Many-Particle Systems - Final Report, January 1,  
 1960 - August 31, 1963,"  
 September 30, 1963, Contract AF61(052)-351

N64-16739-08-25, NASA CR-53102, EE-4012-101-644,  
 R. L. Ramey,  
 "A Study of Thin Film Vacuum Deposited Junctions - (First) Annual  
 Status Report, December 5, 1962 - December 5, 1963,"  
 January, 1964, NASA Grant NSG-340

N64-16774-08-25, ESD-TDR-64-15, AD-429866,  
 B. Lax, et. al.,  
 "Solid State Quarterly Progress Report, Division 8 (October 1 -  
 December 31, 1963,"  
 February 6, 1964, Contract AF19(628)-500

N64-17618-09-25, NRL-MEMO-1432, AD-416457,  
 C. O. Beachem, B. F. Frown, A. J. Edwards,  
 "Characterizing Fractures by Electron Fractography. Part XII:  
 Illustrated Glossary Section I: Quasi-Cleavage,"  
 June, 1963

N64-17876-11-01, AD-431661,  
 H. J. Zimmermann, G. G. Harvey,  
 "Quarterly Progress Report No. 72 (Period Ending November 30, 1963)  
 A Review of the Research Activities on Radio Physics, Plasma  
 Dynamics, and Communication Sciences and Engineering,"  
 January 15, 1964, NASA Grant NSG-496, Contracts DA-36-039-SC-  
 78108 and AT(30-1-1842) and AF19(614)-5992, NSF(G-16526, et. al.

N64-18255-10-25, RADC-TDR-64-6, AD-434762,  
 Queisser, Schroen, Hooper,  
 "Failure Mechanisms in Silicon Semiconductors, Quarterly Report  
 No. 3 (September 1 - November 30, 1963),"  
 March, 1964, Contract AF30(602)-3016

- N64-18259-10-07, ASD-TDR-63-620, AD-427027,  
M. Hoch,  
"Materials Analytical Research. Technical Documentary Report,  
June 15, 1960 - May 31, 1963,"  
November, 1963, Contract AF33(616)-7450
- N64-18376-10-09, AL-TDR-64-29, AD-432171,  
R. M. Burger,  
"Selected Experiments in Silicon Integrated Device Technology  
(Final Report, January, 1963 - February, 1964),"  
March, 1964, Contract AF33(657)-10340
- N64-18592-11-25, R63ELS-74, RADC-TDR-64-35, AD-435469,  
K. K. Reinhartz, V. A. Russell, et. al.,  
"Failure Mechanisms at Surfaces and Interfaces, Second Interim  
Technical Documentary Report, October 1, 1963 - January 1, 1964,"  
March, 1964, Contract AF30(602)-3085
- N64-18638-11-25, NLCO-903,  
P. R. Morris,  
"The Inversion of Pole Figures for Materials Having Orthorhombic  
Symmetry,"  
April 10, 1964, Contract AT(30-1)-1156
- N64-18657-11-25, Report 51, AD-429325,  
J. C. Slater, M. P. Barnett, et. al.,  
"Solid State and Molecular Theory Group, Quarterly Progress Report,"  
January 15, 1964, Contract Nonr-1841(34), NSF Grants 24908 and  
10821, AF19(628)-356
- N64-18880-11-23, PIBMRI-1008-62, AFCRL-62-166, AD-286308,  
T. Tamir, A. A. Qlines  
"Guided Complex Waves. Part I. Fields at an Interface,"  
March 21, 1962, Contract AF19(604)-7499
- N64-18941-11-23, AFCRL-63-325(2), AD-435620,  
A. Kohan,  
"On Determination of the Optical Absorption Coefficient, II,"  
December, 1963
- N64-18960-11-25, NASA-TN-2275,  
R. J. Bacigalupie,  
"Surface Topography of Single Crystals of Face-Centered-Cubic,  
Body-Centered-Cubic, Sodium Chloride, Diamond, and Zinc-Blende  
Structures,"  
April, 1964
- N64-19018-11-25, LA-3043,  
A. C. Larsan, D. T. Cromer,  
"An Integrated Series of Crystallographic Computer Programs II.  
Processing Collected Intensity Data,"  
April 17, 1964, Contract W-7405-eng-36

- N64-19550-12-25, AD-434381,  
A. C. Zittlemayer, K. D. Iyengar, W. M. Block,  
"Studies of the Surface and Corollary Electrical Properties of  
Germanium, Final Report, April 16, 1961 - February 29, 1964,"  
1964, Grant DA-SIG-36-039-61-G5
- N64-19591-12-09, NASA CR-53841,  
D. R. Corley, R. Rosenzweig,  
"Design, Development and Fabrication of a Fast Silicon Power  
Transistor, Summary Report, May 28, 1962 - August 28, 1963,"  
1963, NASA Contract NAS8-5006
- N64-19824-12-23, NASA CR-53878,  
L. J. Frisco,  
"Investigation of the Behavior of Dielectric Materials at High  
Field Strengths in a High Vacuum Environment, First Quarterly  
Report, April 1 - June 30, 1963,"  
July 11, 1963, NASA Contract NAS8-5253
- N64-19939-12-24, NASA CR-53929, JPL-TR-32-515,  
Douglas B. Nash,  
"Nelo Technique for Quantitative SiO<sub>2</sub> Determination of Helical  
Materials by X-Ray Diffraction Analysis of Glass,"  
July 27, 1963, NASA Contract NAS7-100
- N64-20275-13-23, DI-82, 0169-R1, AD-428452,  
Regis M. N. Pelloux,  
"The Analysis of Fracture Surfaces by Electron Microscopy,"  
December, 1963
- N64-20334-13-9, USAELRDL-TR-2333, AD-408131,  
B. Reich, E. B. Hakim,  
"Bulk Reliability Effects in Semiconductor Devices; Current Crowding  
in Transistors,"  
February, 1963
- N64-20397-13-25, RTD-TDR-63-4145, AD-428995,  
F. N. Hodgson, J. E. Katon,  
"Studies in Solid State Mass Spectrometry, Technical Documentary  
Report, July 1, 1962 - July 1, 1963,"  
December, 1963, Contract AF33(616)-8465
- N64-20453-13-23, AD-435880,  
R. A. Smith, et. al.,  
"Research in Materials Sciences and Engineering, Annual Report No.  
3, 1963-1964,"  
April, 1964, Contracts ARPA-SD90; DA-49-020-AMC-0231(X);  
Nonr 184(10);AF04(694)-305; AT(10-1)-1067, et. al.

- N64-20639-13-15, NASA CR-56015,  
E. S. Machlin,  
"A Field Ion Microscope Facility for Solid State and Surface Studies,"  
March 31, 1964, NASA Grant NsG-294-63
- N64-20642-13-15, NASA CR-56039,  
"Solids Mass Spectrometer, Quarterly Technical Progress Report No. 2,  
February 4 - May 3, 1964,"  
1964, NASA Contract NASw-839
- N64-20690-13-24, NASA CR-56103,  
M. K. Testerman, et. al.,  
"Basic Experimental Research Leading to Improved Cold Electron Sources  
and New Types of Velocity Filter Mass Spectrometers, Final Report  
(April 10, 1964),"  
Grant NsG-153-61
- N64-20967-14-09,  
M. F. Goldberg,  
"Physics of Failure in Electronics Vol. 2 (Proc. Second Annual  
Symposium on the Physics of Failure in Electronics),"
- N64-21086-14-07, RTD-TDR-63-4128, AD-428014,  
G. D. Perkins,  
"Mass Spectrometer Output Interpretation, Technical Documentary  
Report, April 1, 1962 - October 31, 1963,"  
October, 1963, Contract AF33(657)-8174
- N64-21148-14-09, SB-64-2, Report 3-35-64-2, AD-600708,  
J. B. Goldmann,  
"Printed Circuits and Welded Modules (1963); An Annotated Bibliography,"  
December, 1963, Contract AF04(647)-673
- N64-21725-15-25, AFCRL-64-360, AD-600799,  
S. P. Wolsky,  
"Investigation of Ultra High Vacuum Sputtered Thin Films, Scientific  
Report No. 1,"  
May, 1964, Contract AF19(628)-3840
- N64-21769-15-25, MEMO-400, AD-428846,  
P. G. Sedlewicz, R. E. Onley, C. R. Kannewurf,  
"Electron and Hole Injection by a Metal-Depletion Layer Contact,"  
December 30, 1963, Contract N0w-62-0749-D
- N64-21832-15-25, RTD-TDR-63-4264, AD-433828,  
E. Krikorian, R. S. Sneed, et. al.,  
"Vacuum Evaporated and Cathodic Sputtered Thin Films,"  
March, 1964, Contract AF33(657)-8728

- N64-21953-15-25, FTD-TT-63-687/1+2, AD-428315,  
E. M. Omel'yanovskiy, et. al.,  
"Mobility of Electrons in Highly Alloyed Silicon,"  
1963
- N64-21967-15-15, FTD-TT-62-1863/1+2+4, AD-421460,  
A. A. Korsuanskiy, I. G. Razumnov,  
"A Thickness Monitor for Thin Films Obtained by Vacuum Spraying,"  
1960
- N64-22222-15-25, ESD-TDR-64-16, AD-435023,  
"Solid State Research, 1963,"  
March 9, 1964, Contract AF19(628)-500
- N64-22278-15-09, RADC-TDR-64-111, AD-600875,  
"Detailed Study of Deleterious Effects on Silicon Transistors,"  
April, 1964, Contract AF30(602)-3244
- N64-22505-15-19, AD-600316,  
S. W. Chaikan, G. A. St. John,  
"Dielectric Thin Films with Increased Breakdown Strength,"  
April 15, 1964, Contract DA-36-039-SC-B7460
- N64-22530-15-25, Q445-7, AD-601021,  
E. Rusmanis, J. Cline, R. Tierman,  
"Active Thin Film Techniques Micromin Program, Interim Development  
Report No. 7, February 1 - April 30, 1964,"  
May 28, 1964, Contract NObsr-87633
- N64-22612-15-23, AFCRL-64-383, AD-441461,  
W. W. Hunt, Jr.,  
"Charge Transfer Studies with a Time-of-Fight Mass Spectrometer: II  
Kinetic Analysis Including Attenuation of Both Neutrals and Ions by  
Scattering, Physical Sciences Research Paper No. 16,"  
May, 1964
- N64-22762-16-07, NASA CR-50678,  
K. Biemann,  
"The Detection and Identification of Organic Matter by Mass  
Spectrometry, Second and Third Semiannual Report, January 1, 1962 -  
May 31, 1963,"  
1963, Grant NsG-211-62
- N64-22897-16-25, ML-1131, AD-431112,  
R. H. Pantell, M. DiDomenico,  
"The Theory of Direct Transitions in Semiconductors,"  
1964, Contract Nonr-225(48)
- N64-23176-16-25,  
M. J. Bloom,  
"Solid State Bonding,"  
1964

- N64-23290-16-19, FTD-TT-62-1709/1+2, AD-401734,  
S. M. Stishov, N. V. Belov,  
"Crystalline Structure of the New Close-Grained Modification of  
Silica, SiO<sub>2</sub>,"  
February 13, 1963
- N64-23883-16-09, SB-62-67, Report 3-35:62-8, AD-410235,  
J. B. Goldmann,  
"Microminaturization; Its Application to Electronic Circuits and  
Components: An Annotated Bibliography,"  
June, 1963, Contract AF04(647)-787
- N64-23957-16-09, RADC-TDR-64-125, AD-601819,  
"Failure Mechanisms in Microelectronics, Third Quarterly Report,"  
May, 1964, Contract AF30(602)-3017
- N64-23960-16-09, RADC-TDR-64-153, AD-601820,  
W. Schroen, W. W. Hooper, H. J. Queisser,  
"Failure Mechanisms in Silicon Semiconductors, Fourth Quarterly  
Report, November 30, 1963 - February 29, 1964,"  
May, 1964, Contract AF39(602)-3016
- N64-24199-16-16, AD-417763,  
R. C. Willson,  
"The Theory, Design, and Development of An Interferometric  
Spectrometer, Scientific Report No. 1 and 2, January, 1962,  
June, 1963,"  
July 1, 1963, Contracts AF19(604)-7240, AF19(628)-287
- N64-24787-17-23, NASA CR-84,  
P. J. Bryant, C. M. Gosselin,  
"Extreme Vacuum Technology (Below 10<sup>-13</sup> Torr) and Associated Clean  
Surface Studies,"  
July, 1964, Contract NASr-63(06)
- N64-25320-17-25, SAR-8, AD-439958,  
L. N. Hadley,  
"Studies, Research, and Investigations at the Optical Properties of  
Thin Films of Metal Semiconductors and Dielectrics, Semi-annual  
Report, February 16 - August 15, 1963,"  
1963, Contract DA-44-009-eng-3828
- N64-25624-17-09, TRI, AD-439341,  
R. A. Horn,  
"Theory of Electrical Conductance in Acidic Aqueous Solutions,"  
May 15, 1964, Contract Nonr-4424(00)

N64-25751-17-23, A046-F, AD-602043,  
J. P. Spatt,  
"Thin Film Active Devices Final Report, June 22 - September 30, 1963,"  
February 20, 1964, Contract DA-49-186-ORD-1056

N64-25976-18-25, AFCRL-64-278, AD-600725,  
E. B. Dale, G. Senecal, et. al.,  
"The Preparation and Properties of Thin Films of Semiconductors,"  
March 1, 1964, Contract AF19(604)-7218

N64-26012-18-09, NAVWEPS-8190, AD-600676,  
G. G. Kretschman, R. F. Potter,  
"Techniques and Apparatus for Evaporating Compound Semiconductors,"  
April 1, 1964

N64-26230-18-25, NOLTR-64-30, AD-442905,  
W. W. Scanlon,  
"Solid State Research for the Year 1963,"  
February 17, 1964

N64-26273-18-25, AD-440163,  
E. L. Lind, E. R. Hill, D. H. Dickey,  
"Study of Photoconductivity in Thin Films, Interim Engineering  
Report, November 1 - May 1, 1963,"  
May 15, 1964, Contract AF33(615)-1090

N64-26264-18-25, ESD-TDR-64-47, AD-439365,  
A. McWhorter, P. E. Tannenwald,  
"Solid State Division 8 Quarterly Progress Report, January 1 -  
March 31, 1964,"  
May 5, 1964, Contract AF19(628)-500

N64-26487-18-07, DP-879,  
J. A. Wheat,  
"Determination of Metallic Impurities in Water by Atomic Absorption  
Spectrometry,"  
June, 1964, Contract AT(07-2)-1

N64-26649-18-25, AD-439686,  
"Solid State Research, 1963, No. 2,"  
August 16, 1963, Contract AF19(628)-500

N64-26898-18-25, AFCRL-64-564, AD-603090,  
P. T. Landsberg, D. A. Evens,  
"Research on Recombination Fluctuation Theory in Semiconductors,  
Scientific Report No. 1, February, 1963 - April 30, 1964,"  
May 31, 1964, Contract AF61(052)-664

- N64-26918-18-25, RADC-TDR-64-138, AD-603279,  
K. F. Greenough,  
"Failure Mechanisms at Metal-Dielectric Interfaces,"  
July, 1964, Contract AF30(602)-3266
- N64-26919-18-25, RADC-TDR-64-168, AD-603262,  
G. Rupprecht, J. Gilbert,  
"Study of Surface States in Semiconductors,"  
July, 1964, Contract AF30(602)-3157
- N64-27216-19-25, AROD-2178-8, AD-443952,  
C. S. Barrett,  
"Diffraction Studies of Crystal Imperfections, Final Report, July 1,  
1961 - June 30, 1964,"  
June 30, 1964, Grant DA-ARO(D)-31-124-G(52)
- N64-27235-19-25, SMIR-63-3, AD-437866,  
E. J. Soxman, G. N. Steele,  
"Thin Film Research,"  
November 4, 1963, Contract Nonr-4165(00)
- N64-27344-19-25, AD-439233,  
"Semiconductor Research, Semiannual Report, October 1, 1963 - March 31,  
1964,"  
1964, Contract DA-31-124-ARO(D)-17
- N64-27425-19-25, ARL-64-98,  
W. R. Rambausk, e,  
"Optical Properties of Semiconducting Crystals,"  
June, 1964, Contract AF33(616)-7500
- N64-27928-20-25,  
P. Wong, B. Selikson, et. al.,  
"Development of Epitaxial Technology for Microelectronics and Large  
Area Device Application, Quarterly Report, February 1 - April 30,  
1964,"  
June 1, 1961
- N64-28579-20-25, AFCRL-64-467, AD-603783,  
H. H. Woodbury, M. Avea, et! al.,  
"Semiconductor Device Concepts, Scientific Report No. 7,"  
May 5, 1964, Contract AF19(628)-329
- N64-28667-20-25,  
J. W. Allen,  
"Theory of Energy Levels of Transition Metal Impurities in Semi-  
conductors,"  
July, 1964

- N64-28765-20-25, ARPA SD-67, AR-4, AD-602012,  
"Materials Research Center 1963-64 Annual Report, Fourth Annual Report,  
May 1, 1963 - January 31, 1964,"  
1964
- N64-28888-20-08, NASA CR-58597,  
J. Lederberg,  
"Computation of Molecular Formulas for Mass Spectrometry,"  
1964, Grants NsG-81-60, NSF G-6411, NIH G-NB-04270-01,  
NIH G-NB-04270-02, NIH G-AI-5160-06
- N64-28961-20-10,  
M. Lauriente, C. A. Harper, et. al.,  
"Planar Integration of Thin Film Units, First Quarterly Progress  
Report, January 15 - April 14, 1964,"  
1964
- N64-29186-21-07, NASA CR-58503,  
A. E. Hultquist, M. E. Sibert,  
"Electrolytic Preparation of High Dielectric Thin Films, Annual  
Report, June 27, 1963 - June 27, 1964,"  
1964, Contract NASw-765
- N64-29563-21-09, NASA CR-56796,  
M. K. Testerman, B. E. Gilliland, E. M. King,  
"A Cold Electron Mass Spectrometer Ion Source,"  
1964, Grant NsG-260-62
- N64-29398-21-23, SAR-9, AD-445110,  
L. N. Hadley,  
"Studies, Research, and Investigations of the Optical Properties  
of Thin Films of Metals, Semiconductors and Dielectrics, Semi-  
annual Report, August 16, 1963 - February 15, 1964,"  
1964, Contract DA-44-009-eng-3828
- N64-29700-21-15, NASA CR-58538,  
"Solids Mass Spectrometer, Quarterly Technical Progress Report No. 3,  
May 4 - August 3, 1964,"  
1964, Contract NASw-839
- N64-29929-21-25, AMRL-TDR-63-117, AD-604419,  
K. K. Knaell, M. O. Thurston,  
"Investigation of New Semiconductor Phenomenon,"  
June, 1964, Contract AF33(606)-8384
- N64-29967-21-25, RADC-TDR-64-197, AD-604349,  
K. K. Reinhantz, et. al.,  
"Failure Mechanisms at Surfaces and Interfaces, Third Interim  
Technical Documentary Report, January 1 - April 1, 1964,"  
July, 1964, Contract AF30(602)-3085

N64-30006-21-25, AROD-3330-4, AD-604660,

R. K. Gereth, et. al.,

"Research on the Influence of Surface Conditions on Diffusion in Silicon, Final Report, June 16, 1961 - June 15, 1964,"  
July, 1964, Contract DA-04-200-ORD-1166(X)

N64-30123-21-25, AFCRL-64-581, AD-605566,

K. D. Kang, F. Lee, H. W. Cooper,

"Research and Development for Semiconductor Surface Control and Stabilization, Scientific Report No. 1,"  
May 14, 1964, Contract AF19(628)-3808